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# Cathodes for Li-Ion Batteries: Challenges and Prospects

Artem Abakumov

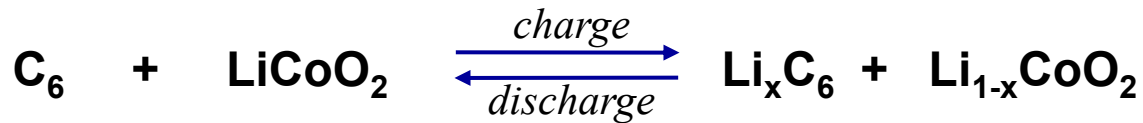
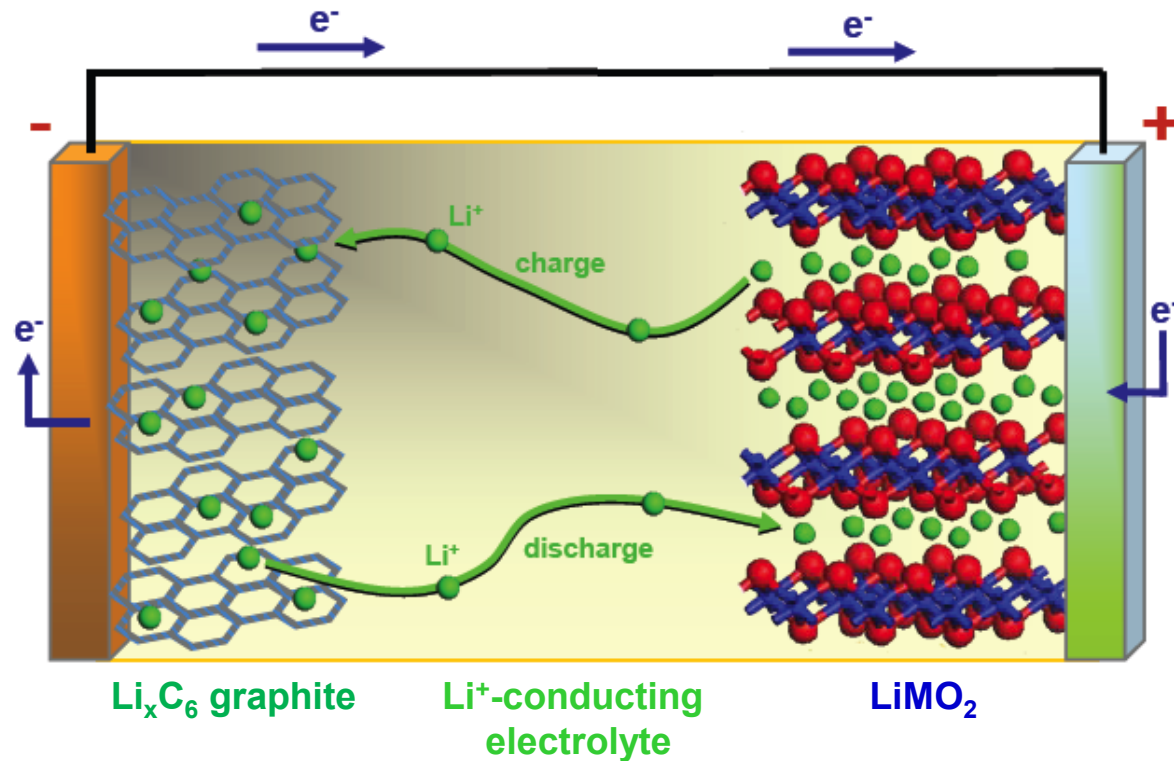
Center for Electrochemical Energy Storage, Skoltech

# Outline

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- 1. Li-ion batteries**
- 2. Cathode materials: key parameters and structures**
- 3. Layered Li-rich cathodes: lattice oxygen redox**
- 4. Layered Li-rich cathodes: cation migration and voltage fade**
- 5. Polyanion cathodes: adjusting the redox potential**
- 6. Conclusions**

# Li-ion batteries



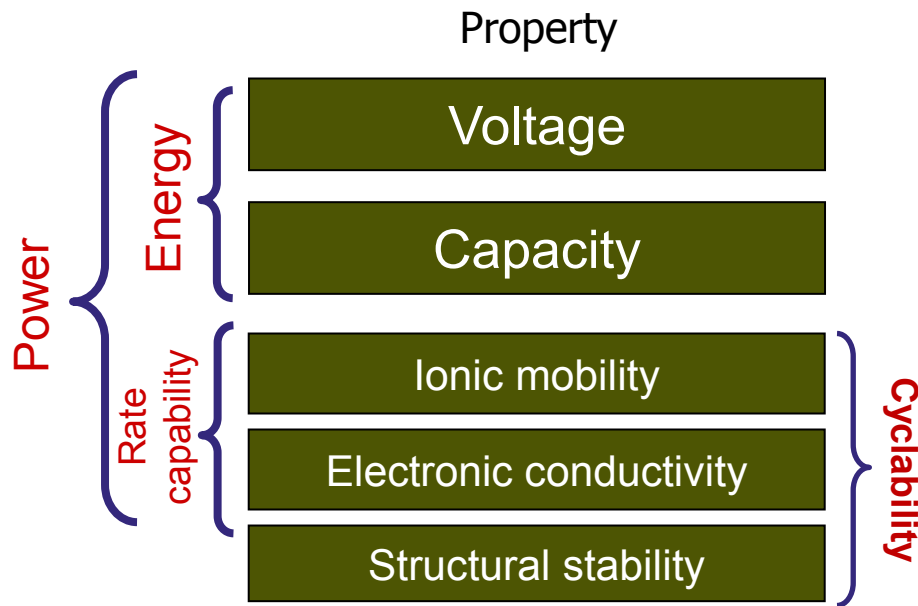
Voltage 3.6 V,  $x \approx 0.5-0.6 e^-$

Electrolyte:

Li-salt -  $\text{LiPF}_6$ ,  $\text{LiBF}_4$  ( $\text{LiClO}_4$ ,  $\text{LiAsF}_6$ ),  $\text{LiCF}_3\text{SO}_3$

Solvent – ethylene carbonate  $(\text{CH}_2\text{O})_2\text{C}$ , dimethyl carbonate  $(\text{CH}_3\text{O})_2\text{CO}$  ....

# Cathode materials: key properties



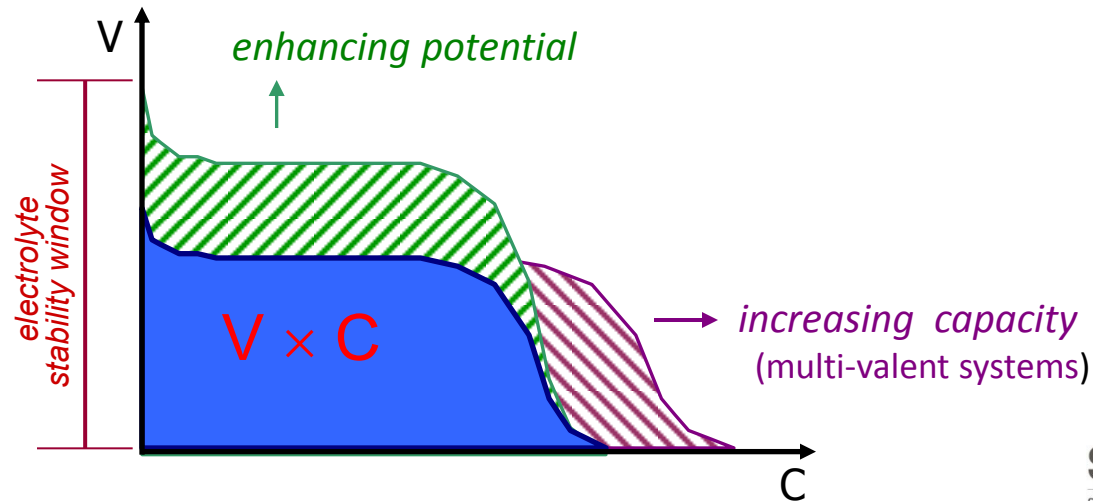
$M^{n+}/M^{(n+1)+}$  redox potential

$$C_T (\text{A h g}^{-1}) = \frac{26.8 \times \Delta n}{M}$$

$\nearrow$  number of  $e^-$  or  $\text{Li}^+$   
 $\searrow$  Molecular weight (g)

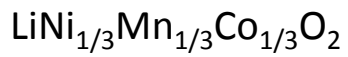
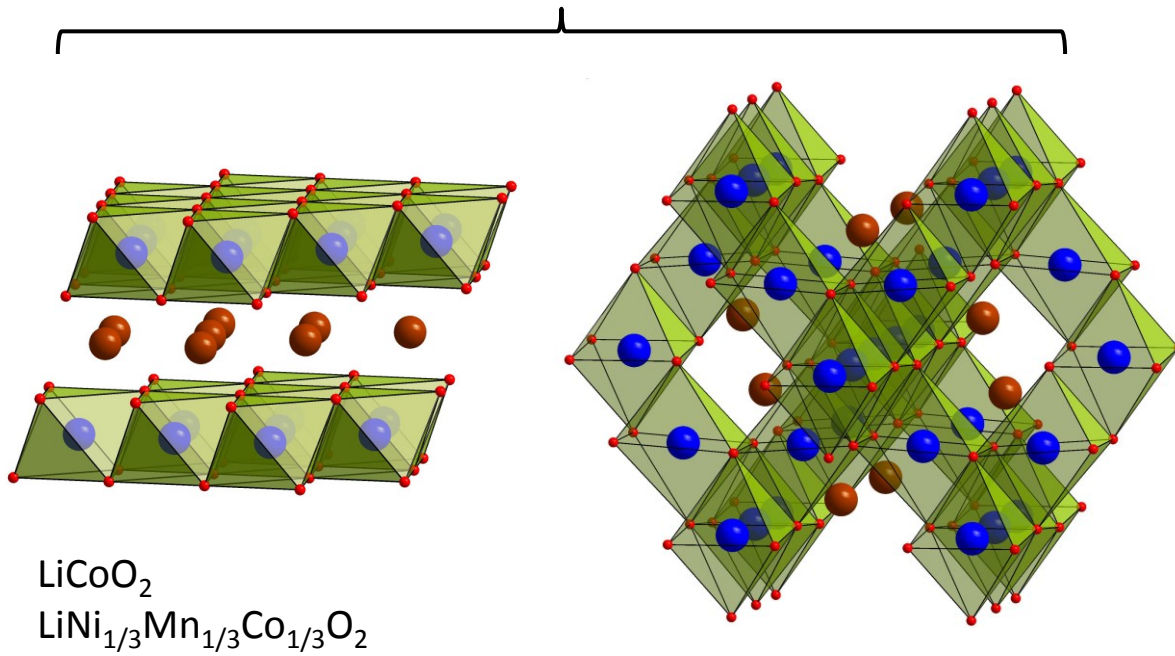
Energy = Voltage x Capacity

$\leq 4.8 \text{ V (vs. Li/Li}^+)$



# Cathode materials

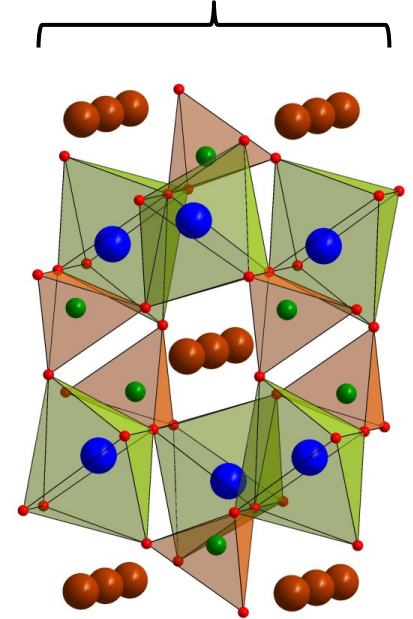
## Complex oxides



2D Li transport

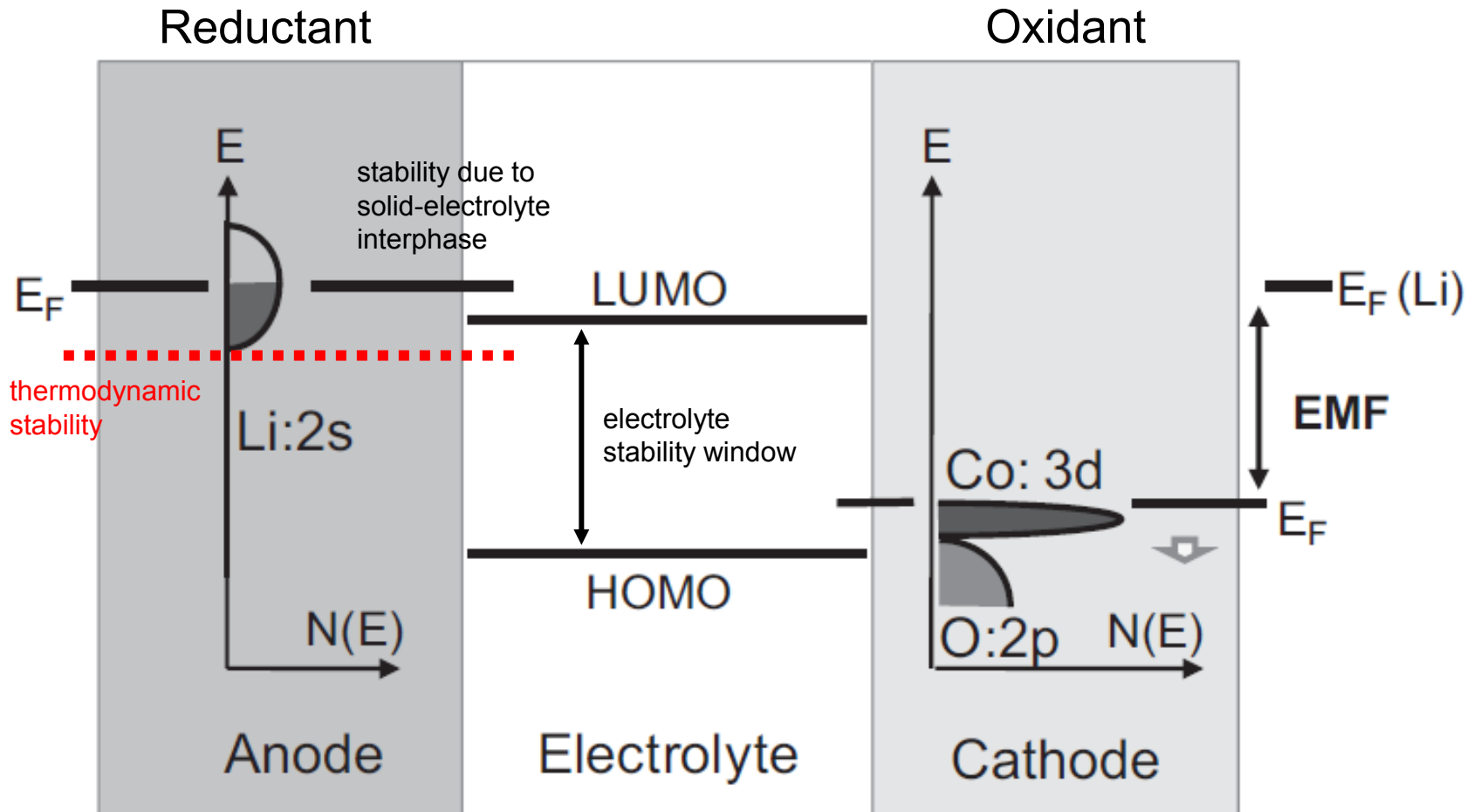
3D Li transport

## Polyanion compounds



1D Li transport

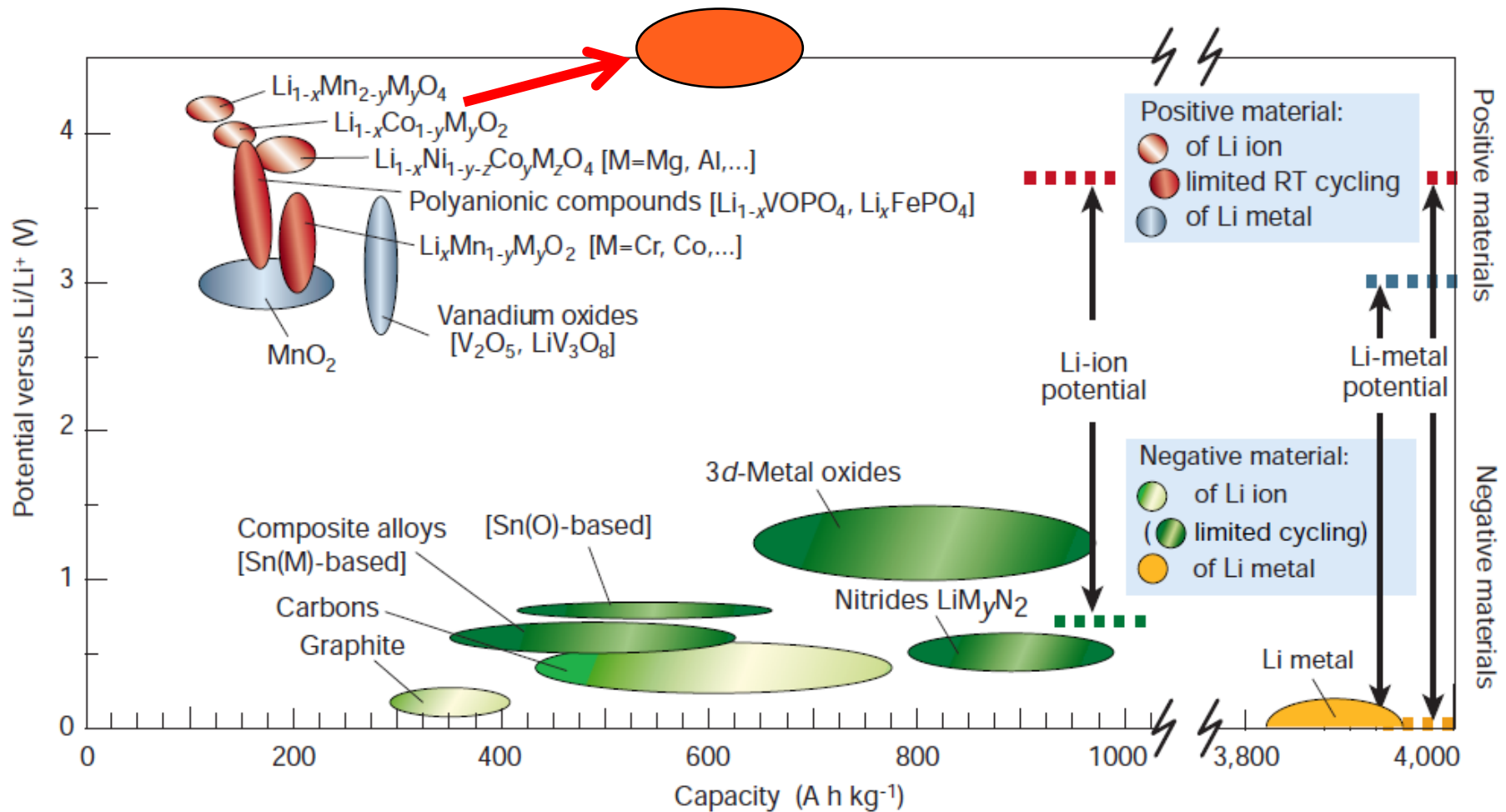
# Li-ion battery energy diagram



# Cathode materials

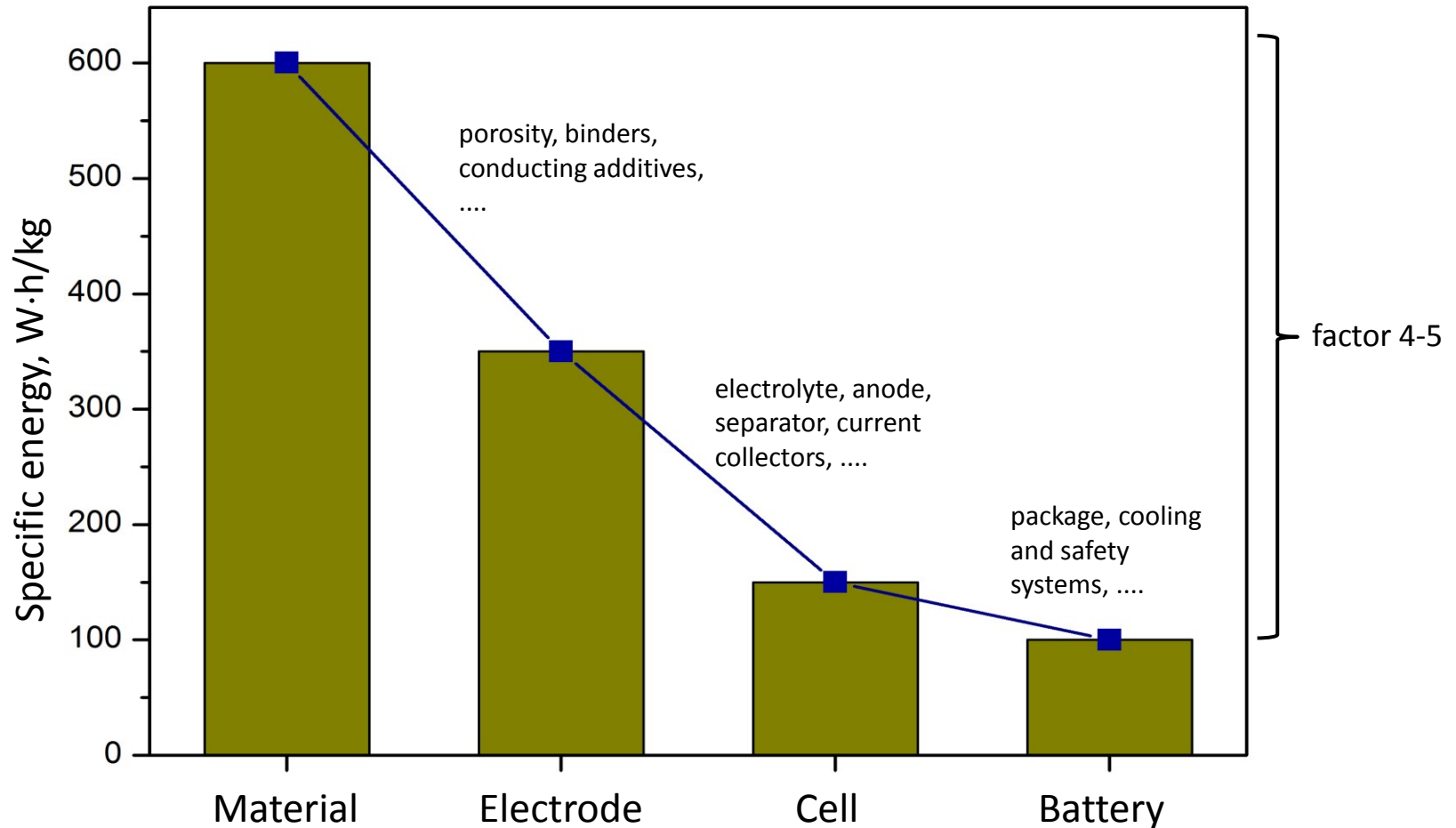
Cathode	LCO	LNO	NCA	NMC	LMO	LFP
Formula	$\text{LiCoO}_2$	$\text{LiNiO}_2$	$\text{LiNi}_{0.85}\text{Co}_{0.1}\text{Al}_{0.05}\text{O}_2$	$\text{LiNi}_{1/3}\text{Mn}_{1/3}\text{Co}_{1/3}\text{O}_2$	$\text{LiMn}_2\text{O}_4$	$\text{LiFePO}_4$
Average potential vs $\text{Li}^+/\text{Li}$ , V	3.7	3.6	3.65	3.9	4.0	3.5
Capacity, mA h/g	~150	~180	~130	~170	~110	~150
Specific energy, W·h/kg	~550	~650	~480	~660	~440	~500
Power	+	0	+	0	+	+
Safety	-	0	0	0	+	++
Life time	-	0	+	0	0	+
Cost	--	+	0	0	+	+

# Cathode and anode materials



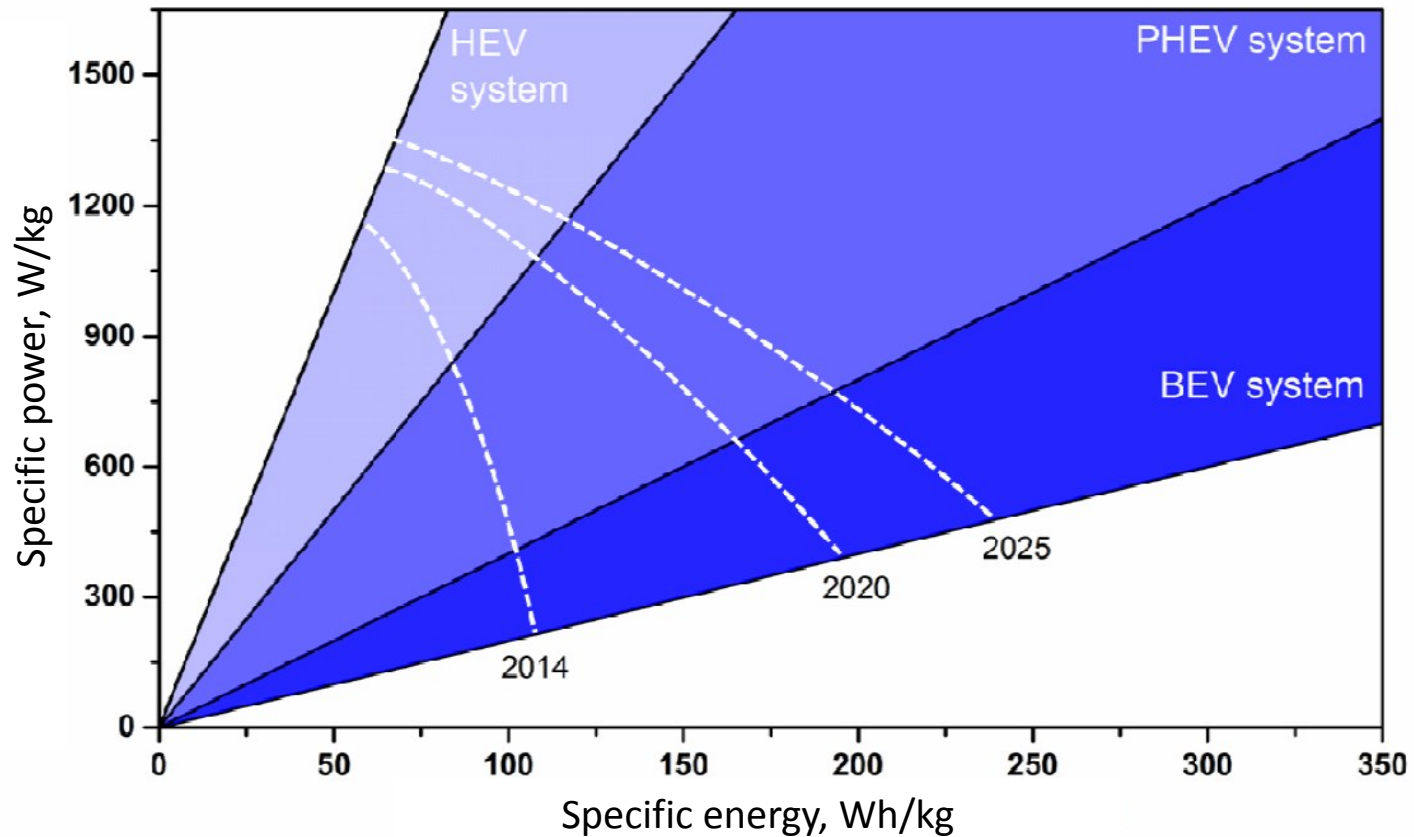


# From cathode material to battery



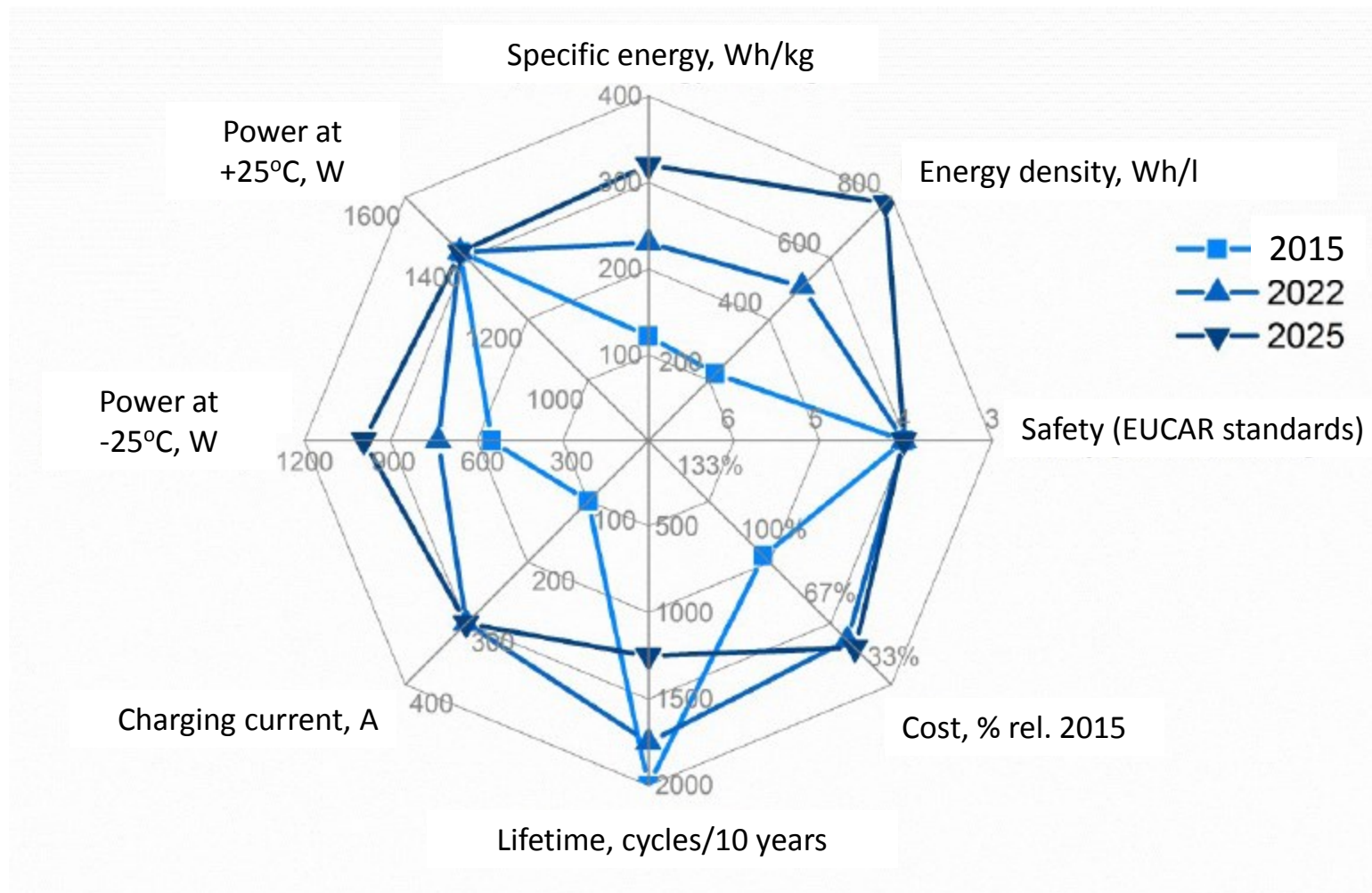
Battery with specific energy of **200 W·h/kg** → cathode with specific energy of **800 – 1000 W·h/kg**

# Batteries for automotive applications



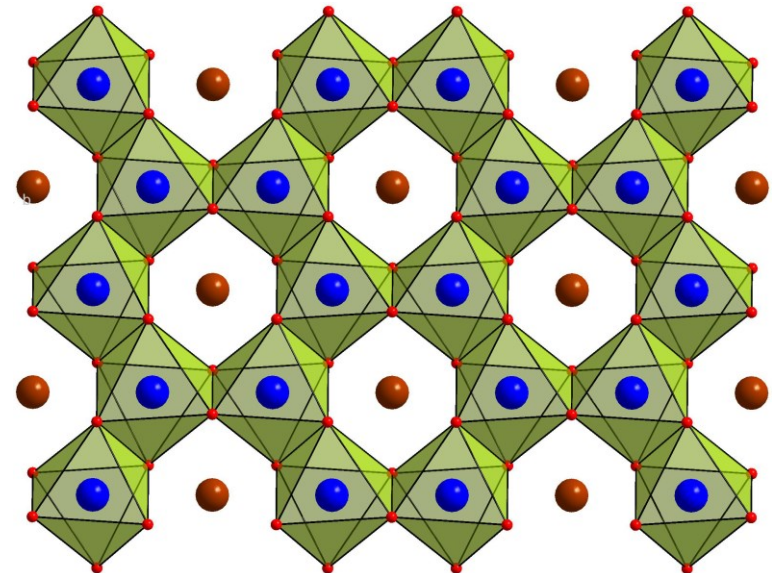
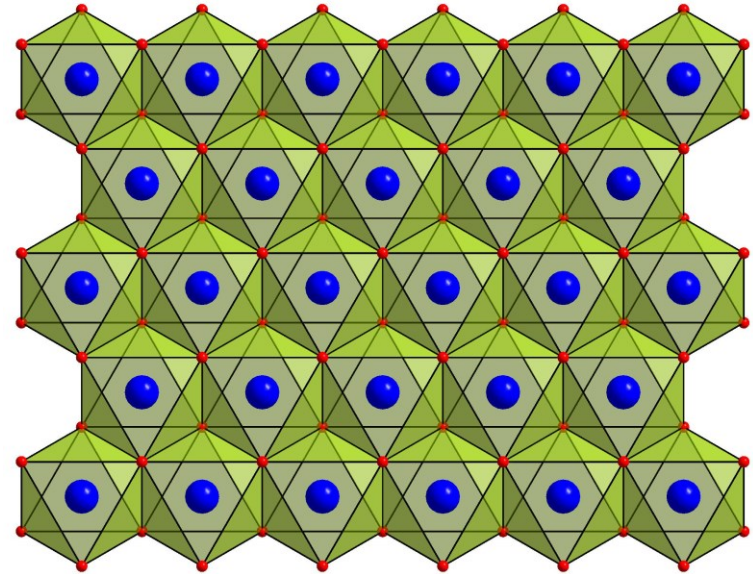
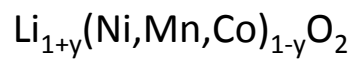
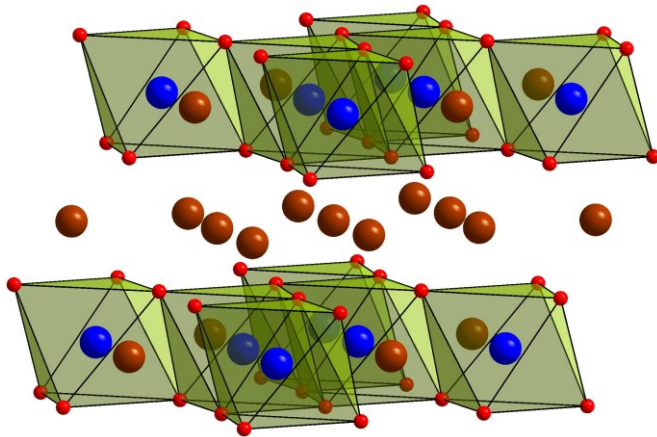
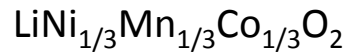
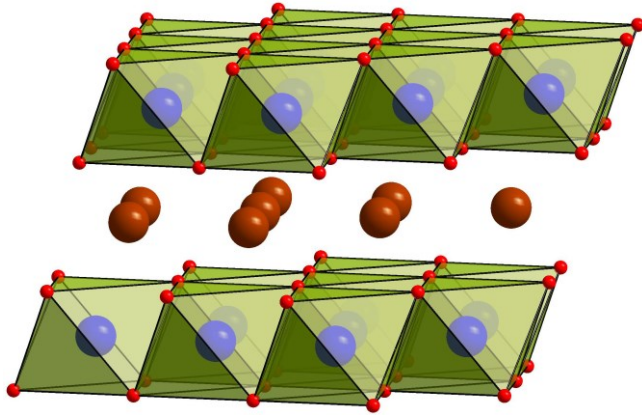
Energy-power diagram for different types of electric vehicles and prospects for 2020 – 2025

# Эволюция основных параметров

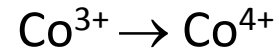
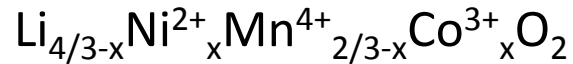


Key performance parameters road map on cell level for fully electrified vehicles from today to 2025.

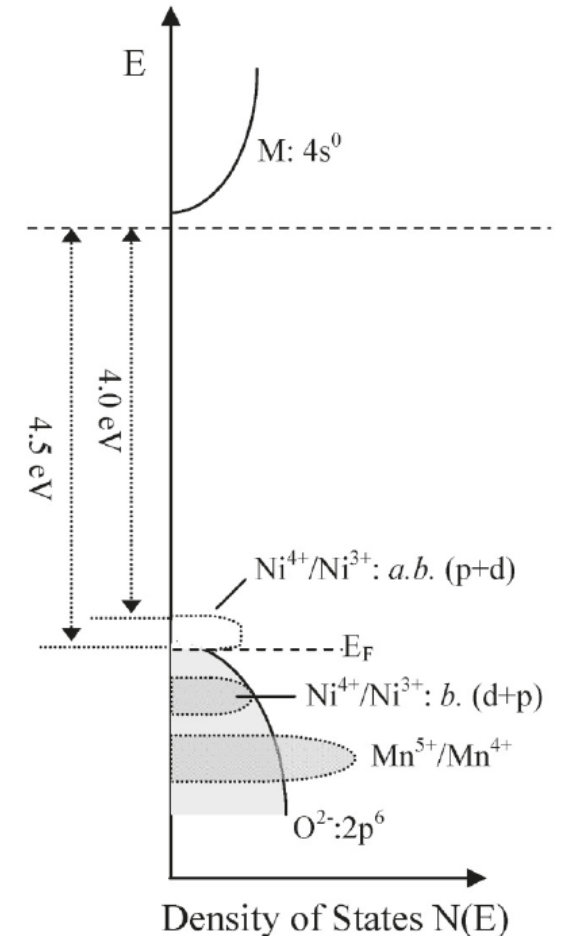
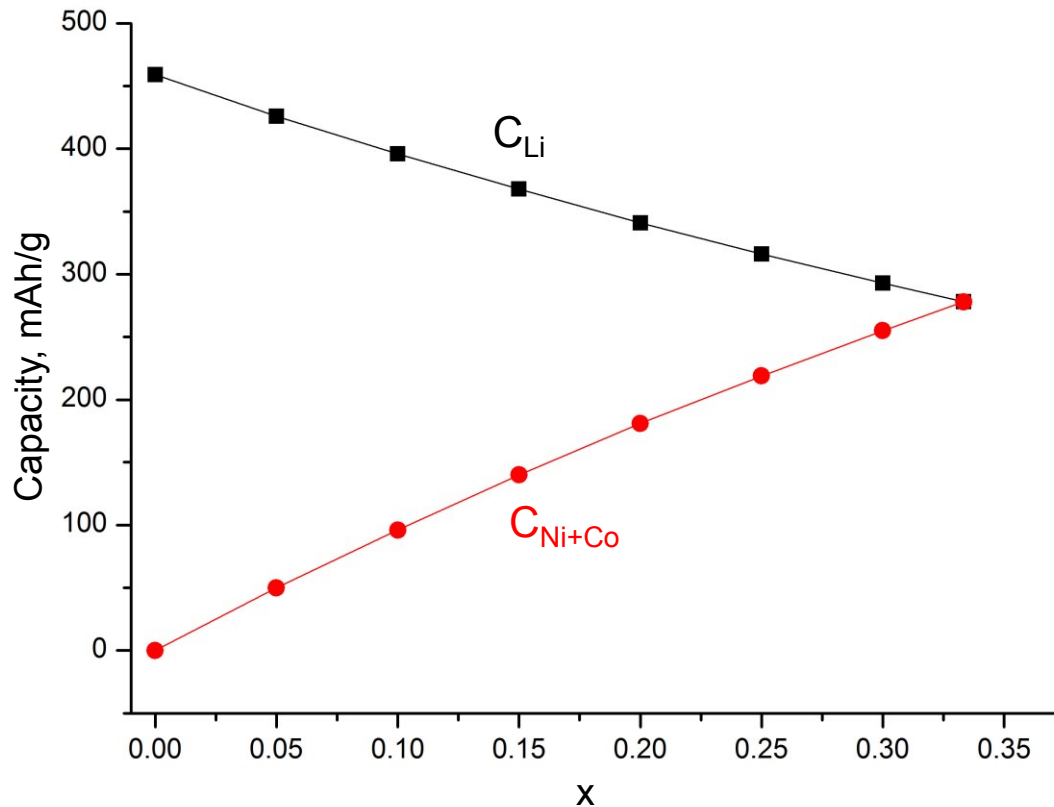
# High capacity layered cathodes



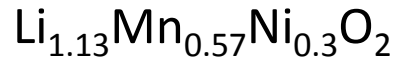
# High capacity layered cathodes



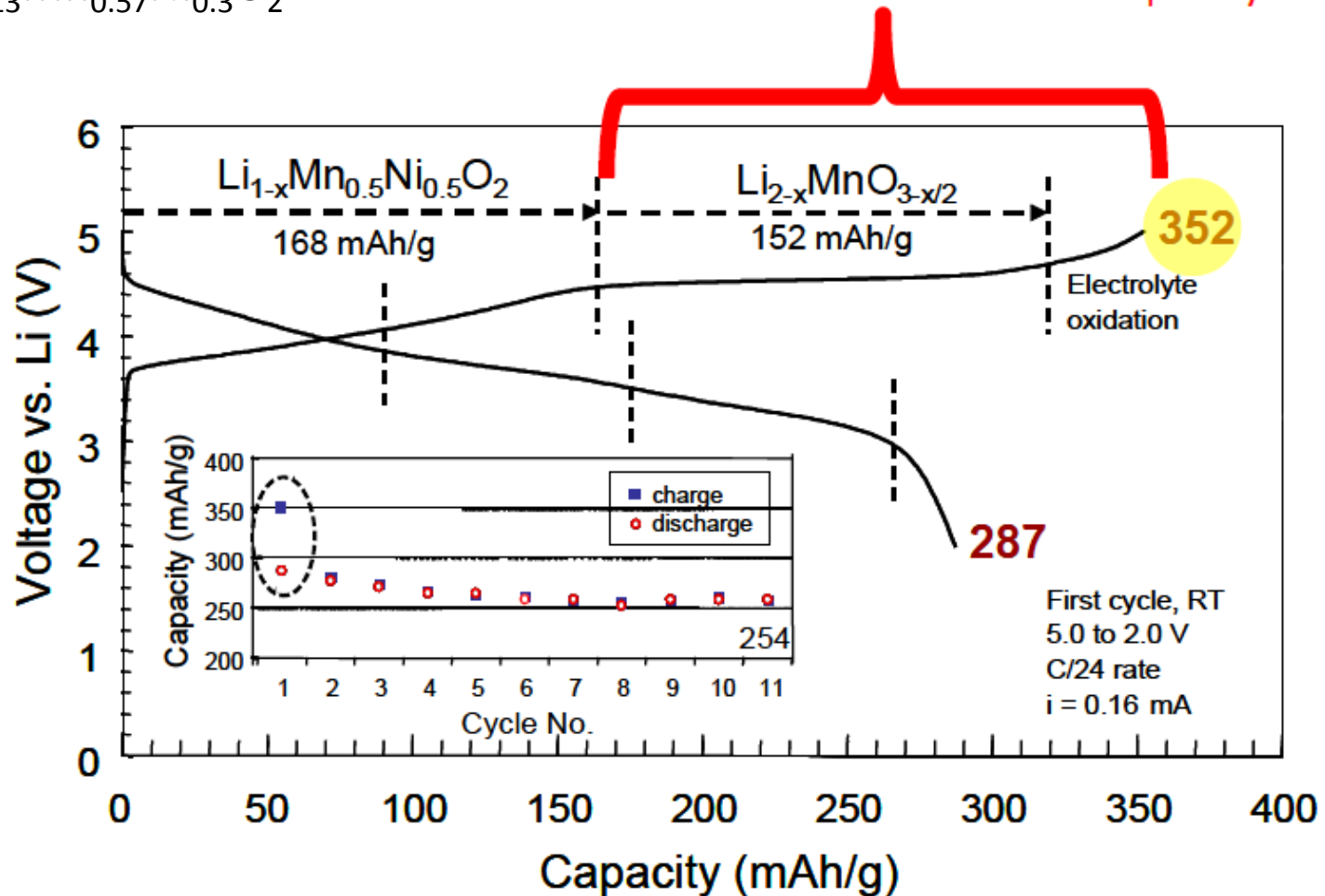
Mn<sup>4+</sup>/Mn<sup>5+</sup> redox couple is inaccessible



# High capacity layered cathodes: excess capacity



What is the cause of the excess capacity?

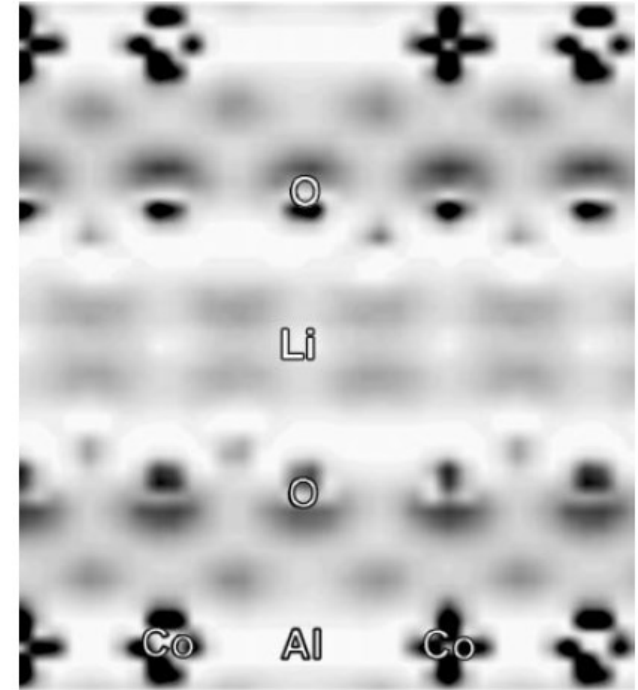


# Lattice oxygen oxidation

## Identification of cathode materials for lithium batteries guided by first-principles calculations

G. Ceder, Y.-M. Chiang, D. R. Sadoway, M. K. Aydinol, Y.-I. Jang & B. Huang

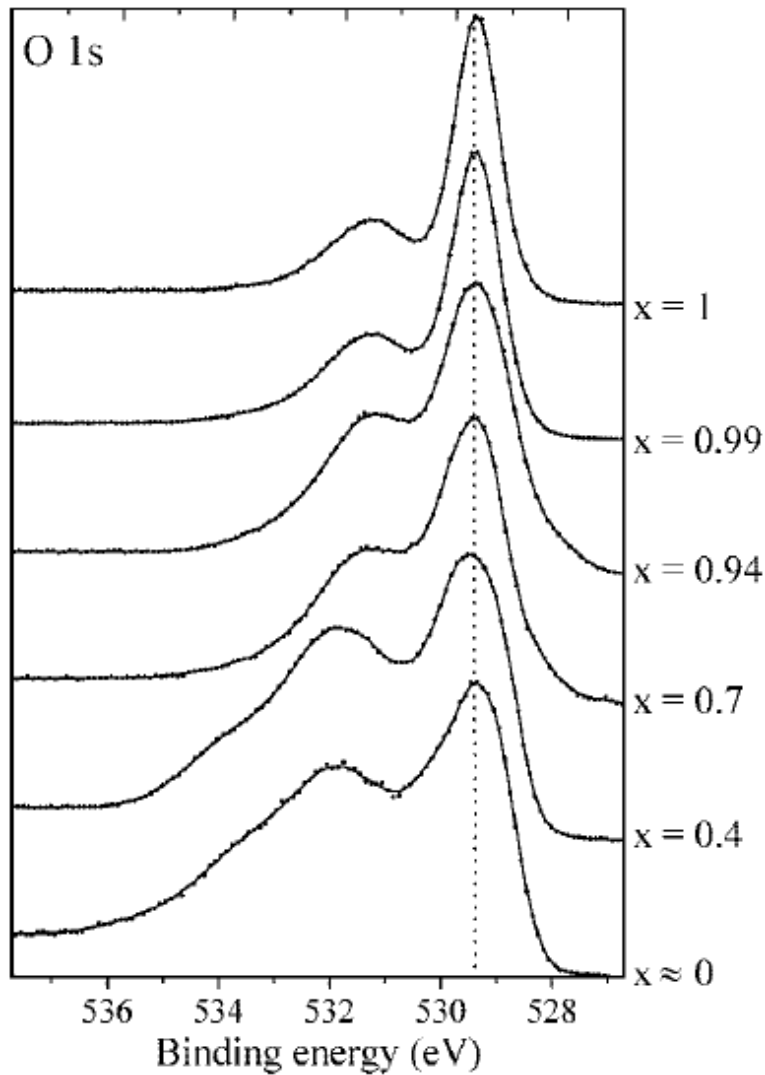
The replacement with non-transition metals is driven by the realization that oxygen, rather than transition-metal ions, function as the electron acceptor upon insertion of Li.



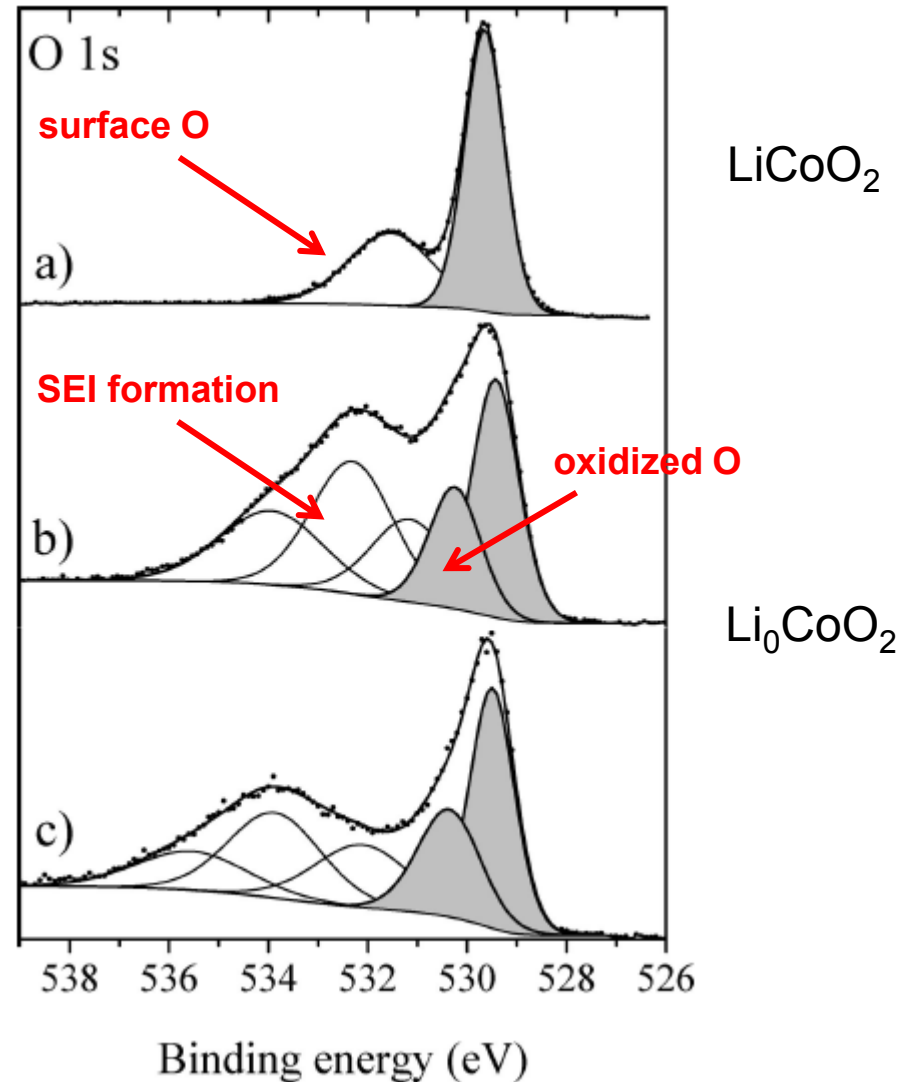
**Figure 1** Positive part of the electron density difference between  $\text{Li}(\text{Al}_{0.33}\text{Co}_{0.67})\text{O}_2$  and  $(\text{Al}_{0.33}\text{Co}_{0.67})\text{O}_2$  in a plane perpendicular to the direction of layering in the structure. Darker indicates larger electron density.

NATURE | VOL 392 | 16 APRIL 1998

# Lattice oxygen oxidation

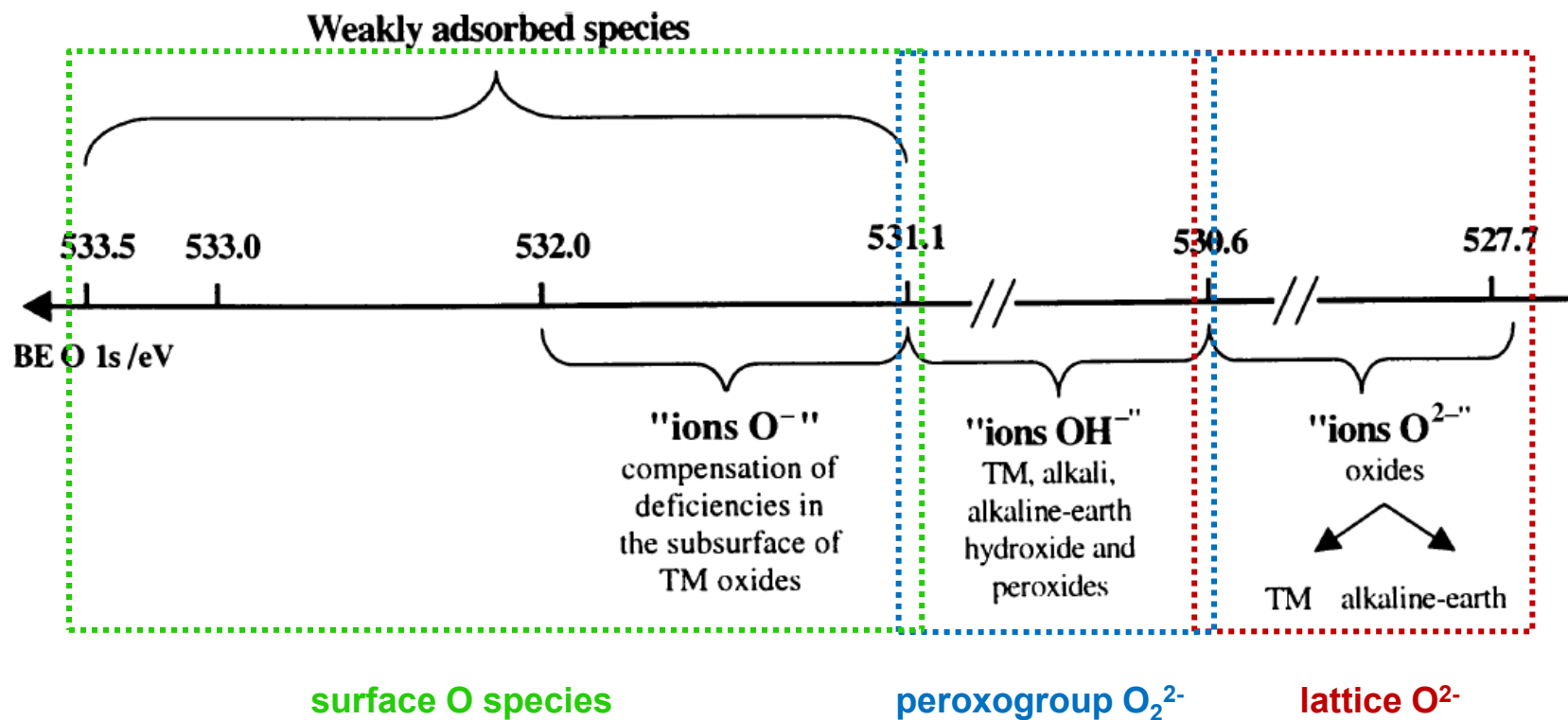


$\text{Li}_x\text{CoO}_2$  XPS O1s

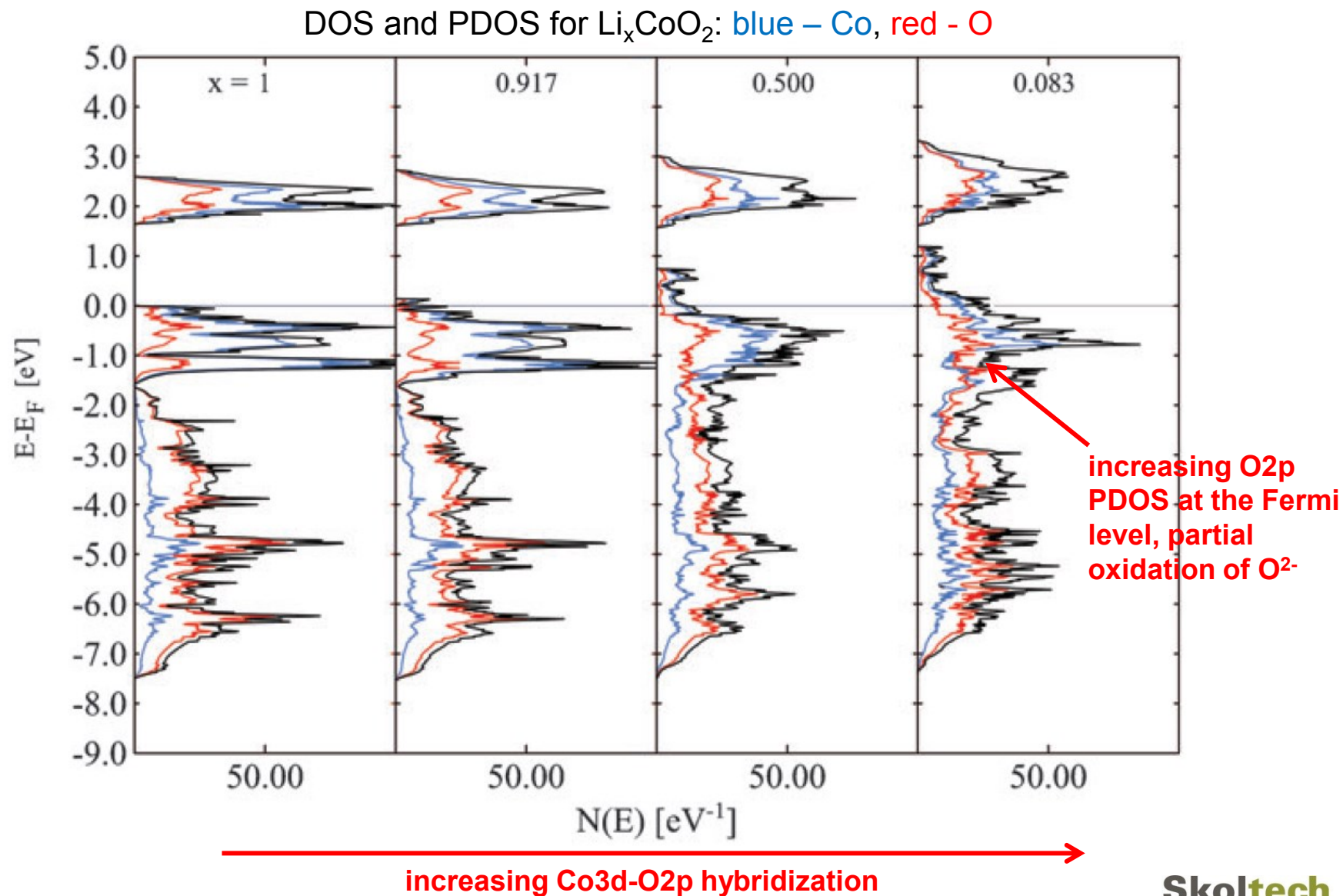




# Lattice oxygen oxidation

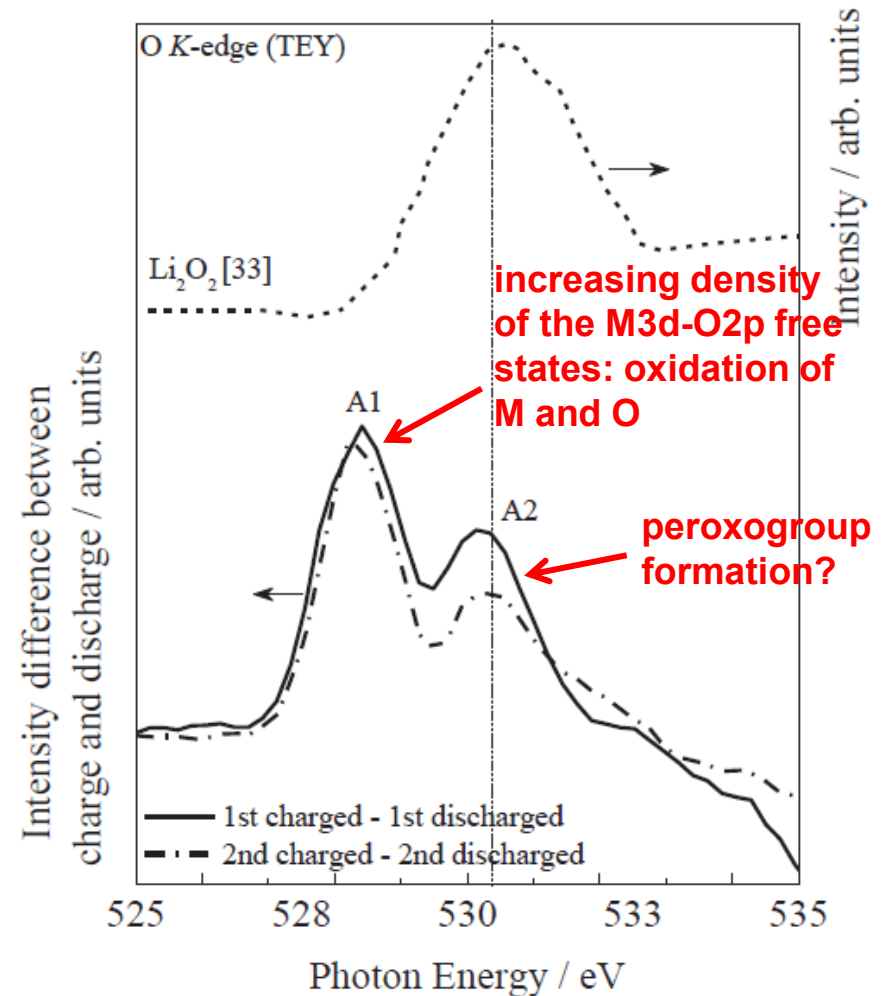
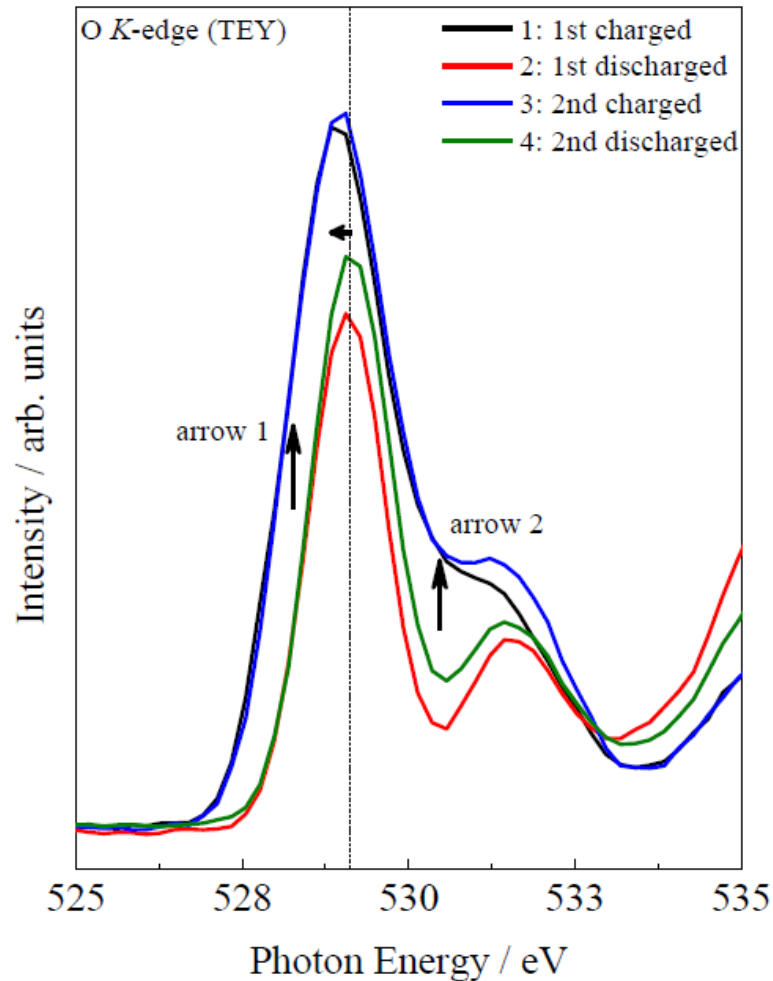


# Lattice oxygen oxidation

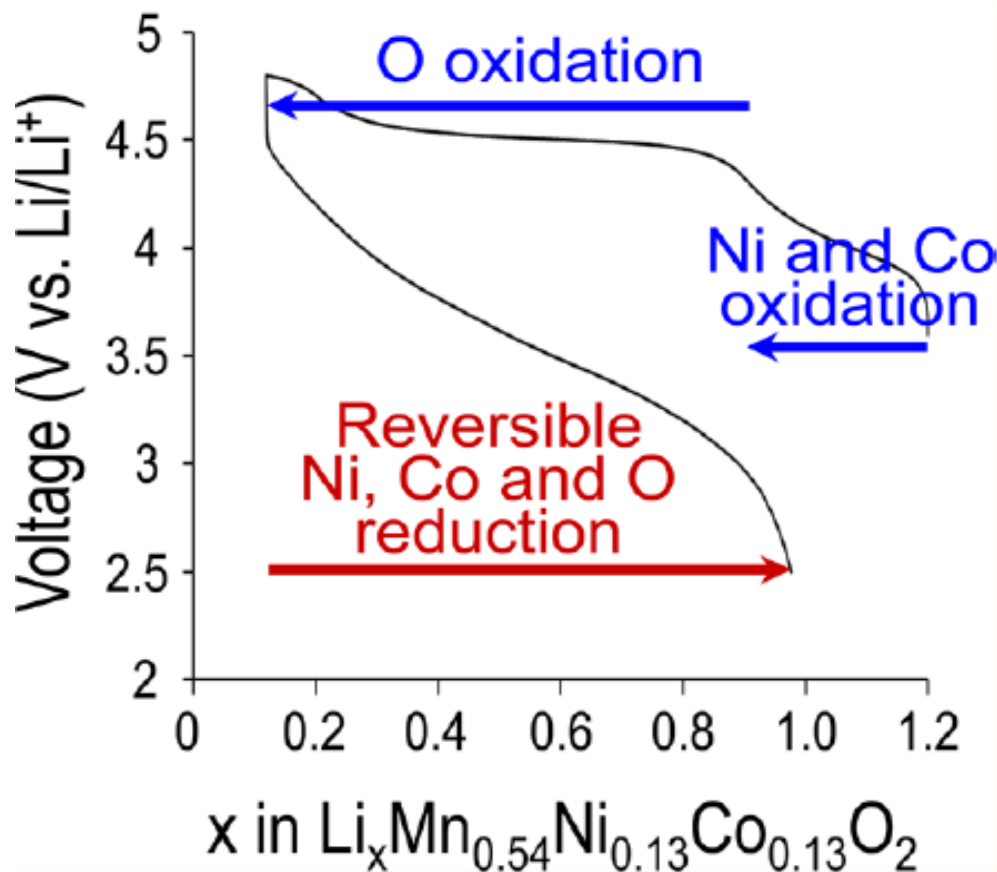


# Lattice oxygen oxidation

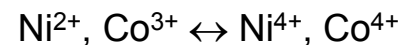
## Li-rich NMC: O-K edge XAFS



# Lattice oxygen oxidation



Two redox processes:

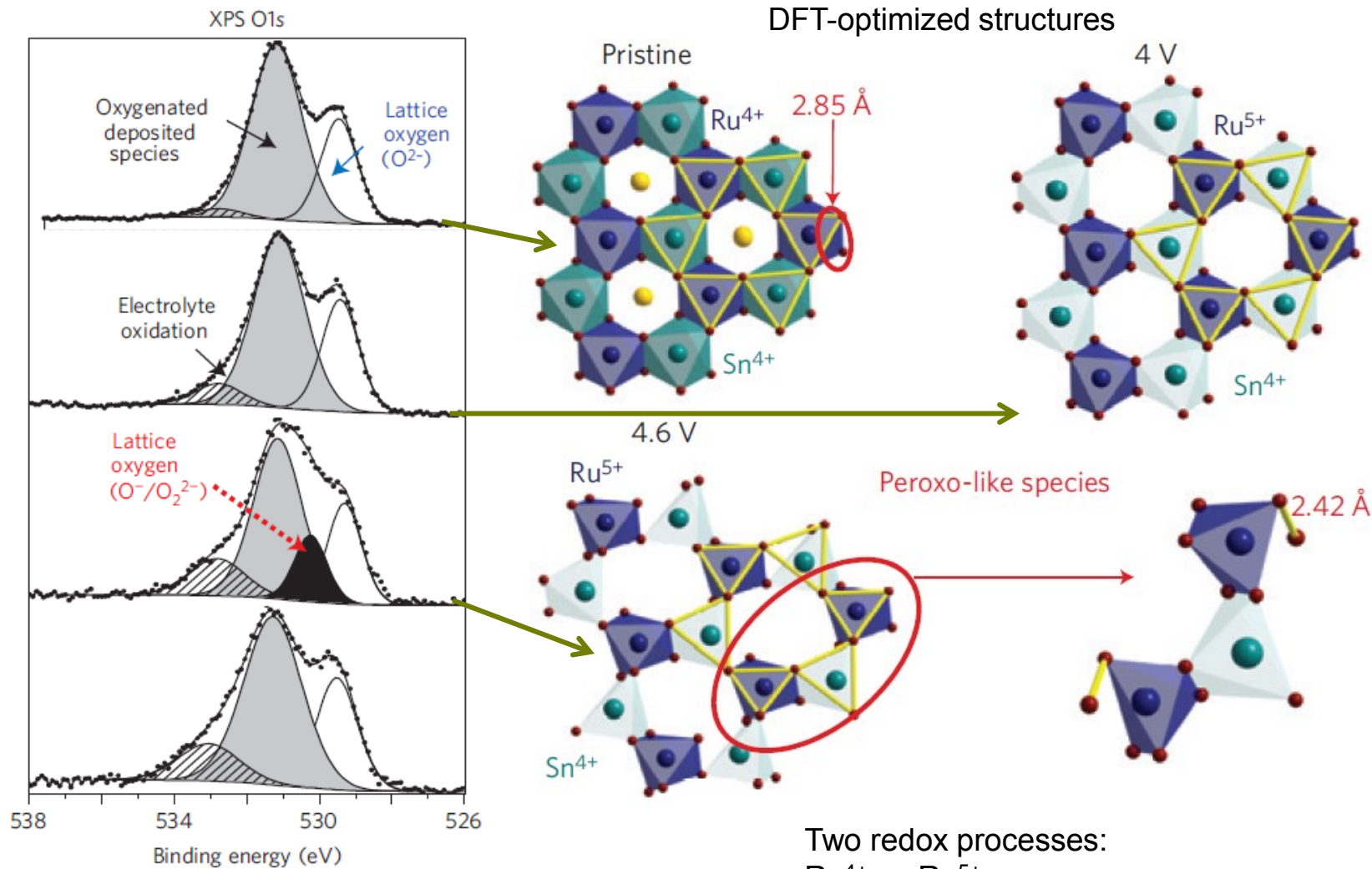
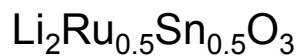


(Mn<sup>4+</sup> is neither oxidized nor reduced)

Reversible oxygen oxidation

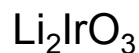
XANES and EXAFS on Ni,Co and Mn-K edges

# Formation of $O_2^{n-}$

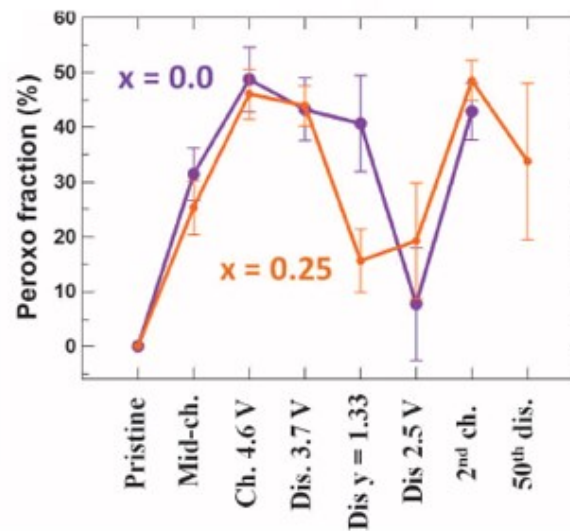
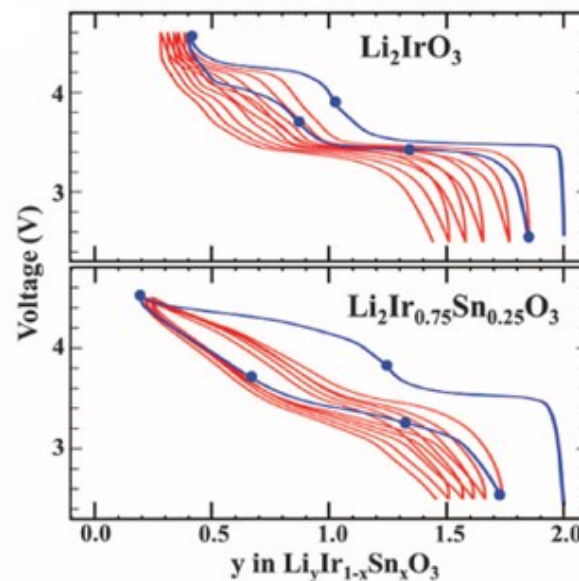
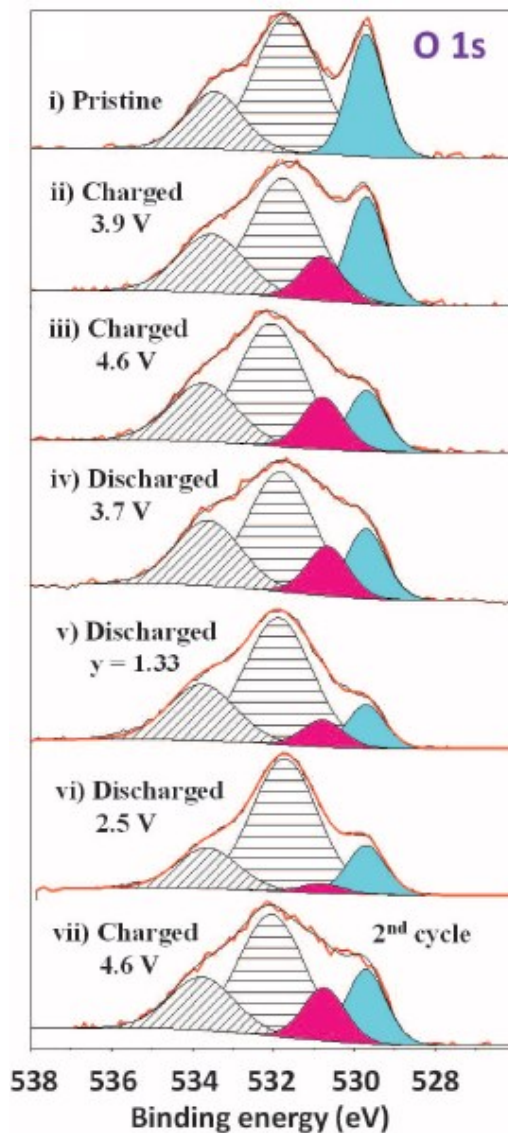
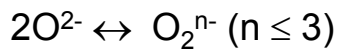


Two redox processes:  
 $Ru^{4+} \leftrightarrow Ru^{5+}$   
 $2O^{2-} \leftrightarrow O_2^{n-} (n \leq 3)$

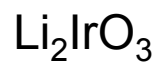
# Formation of $O_2^{n-}$



Two redox processes:

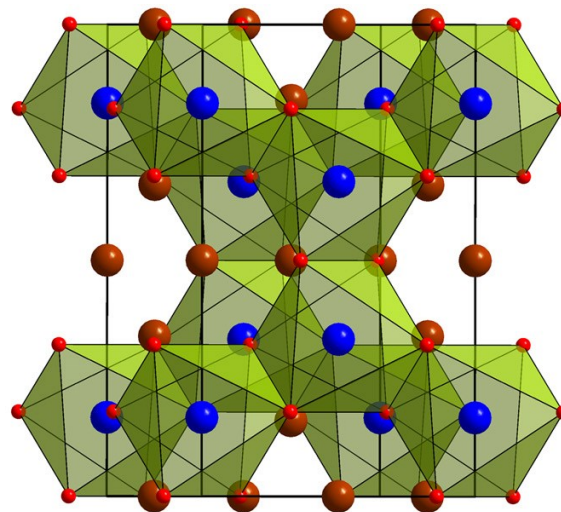
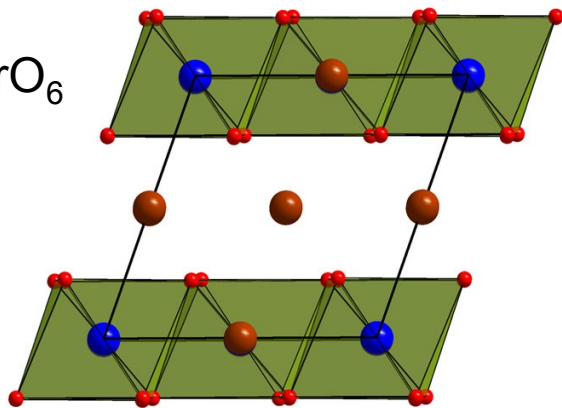


# Formation of $O_2^{n-}$

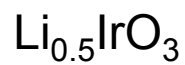


$LiO_6, IrO_6$

Li

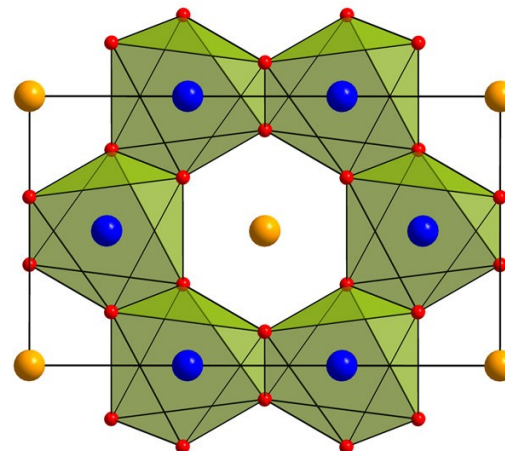
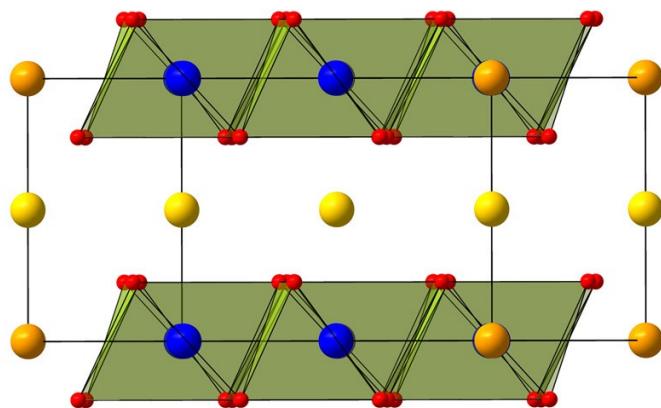


O3



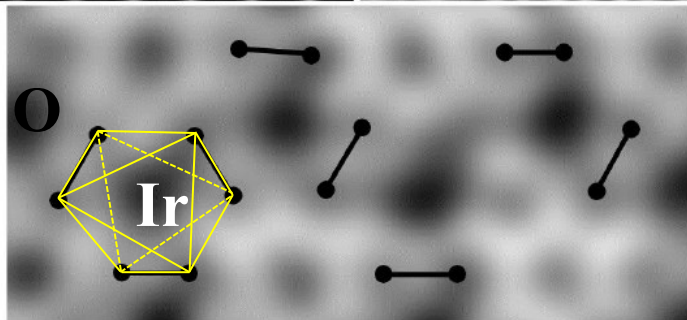
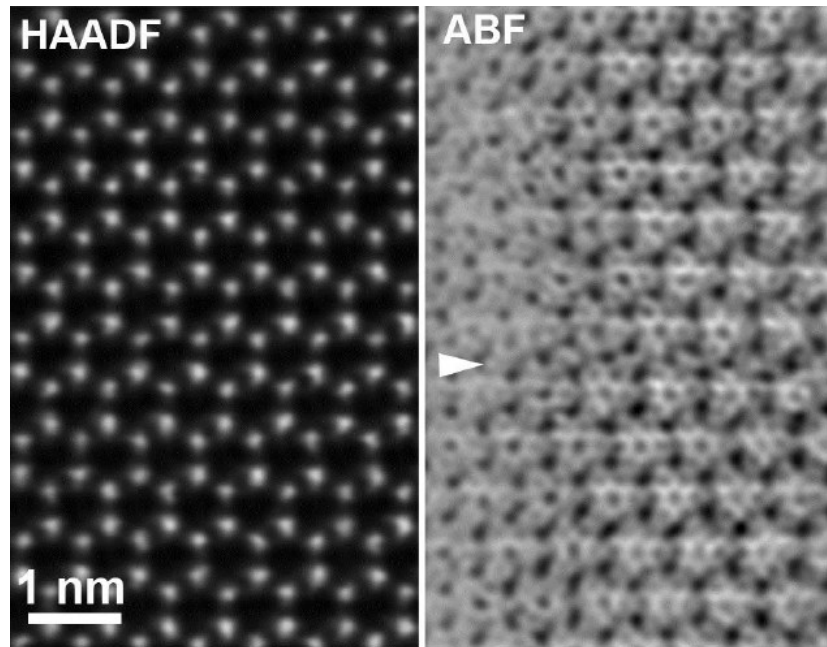
$IrO_6$

$Li_x$

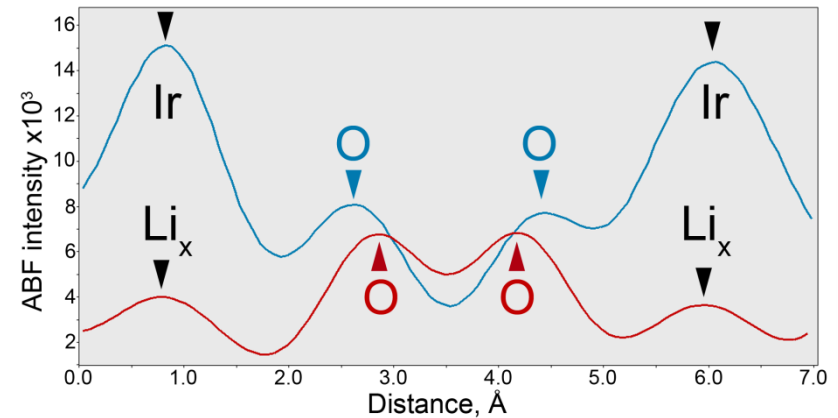


O1

# Formation of $O_2^{n-}$



$Li_2IrO_3 \rightarrow Li_{0.5}IrO_3$  : oxidation of  $Ir^{4+} \rightarrow Ir^{5+}$  and  $O^{2-} \rightarrow O_2^{n-}$  ( $n < 4$ ), shortening the O-O distances



Projected O-O distances from ABF-STEM:

short: 1.56(1)Å long: 1.83(1)Å

Projected O-O distances from DFT ( $Li_{0.5}IrO_3$ ):

short: 1.48Å long: 1.85Å

HAADF- and ABF-STEM for  $Li_{0.5}IrO_3$  charged to 4.5V

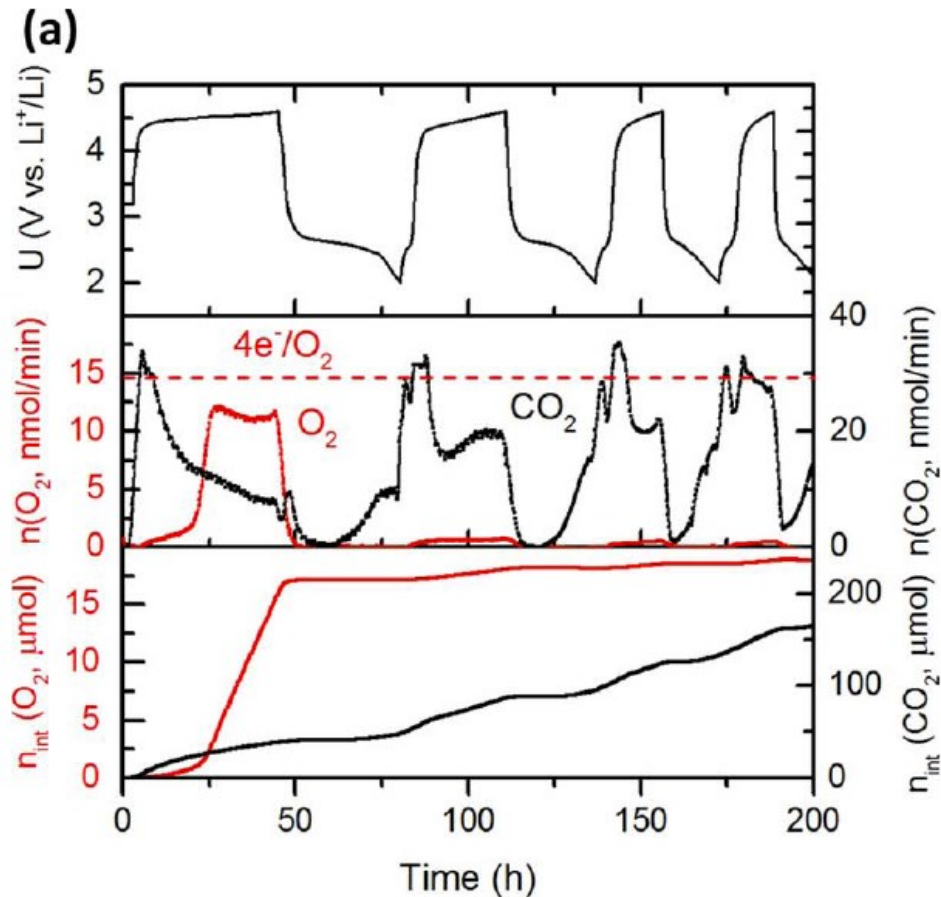


# Formation of $O_2^{n-}$

**Table 1. Average O-O distances obtained by DFT, NPD, and TEM.** “Short” refers to two oxygen atoms between two nearest-neighbor Ir atoms, as viewed in the [001] projection in Fig. 3, E and F. “Long” refers to distances at which the oxygen atoms lie between an Ir atom and a vacancy. In all cases, the distances are averages for the structure. Projected distances are shown for the O1 structure only. N/A, not applicable; ND, not determined.

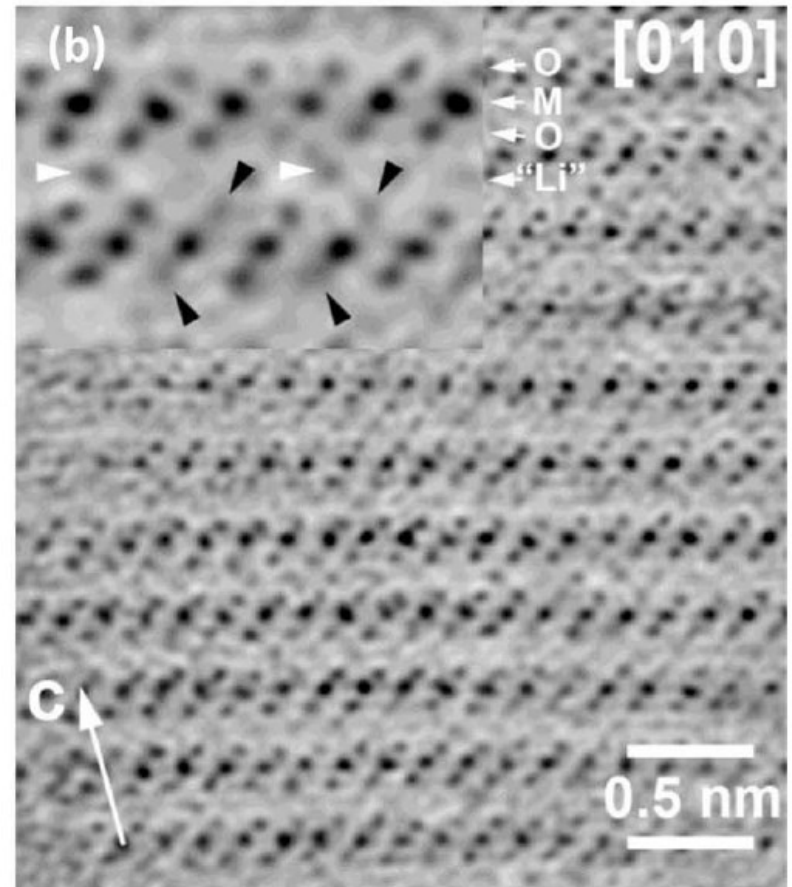
Sample	O-O distance (Å)		O-O distance in [001] projection (Å)	
	Short	Long	Short	Long
<b>Li<sub>2</sub>IrO<sub>3</sub></b>				
Neutron	2.77(2)	2.84(2)	N/A	N/A
DFT	2.74	2.89	N/A	N/A
<b>Li<sub>0.5</sub>IrO<sub>3</sub></b>				
Neutron	2.45(2)	2.73(4)	1.42(1)	1.86(3)
DFT	2.54	2.77	1.51	1.88
TEM	ND	ND	1.56	1.83
LiNi <sub>1/3</sub> Mn <sub>1/3</sub> Co <sub>1/3</sub> O <sub>2</sub> *	2.686	2.686	N/A	N/A
Li <sub>0.04</sub> Ni <sub>1/3</sub> Mn <sub>1/3</sub> Co <sub>1/3</sub> O <sub>2</sub> *	2.553	2.553	N/A	N/A

# Oxygen evolution and thermal runaway



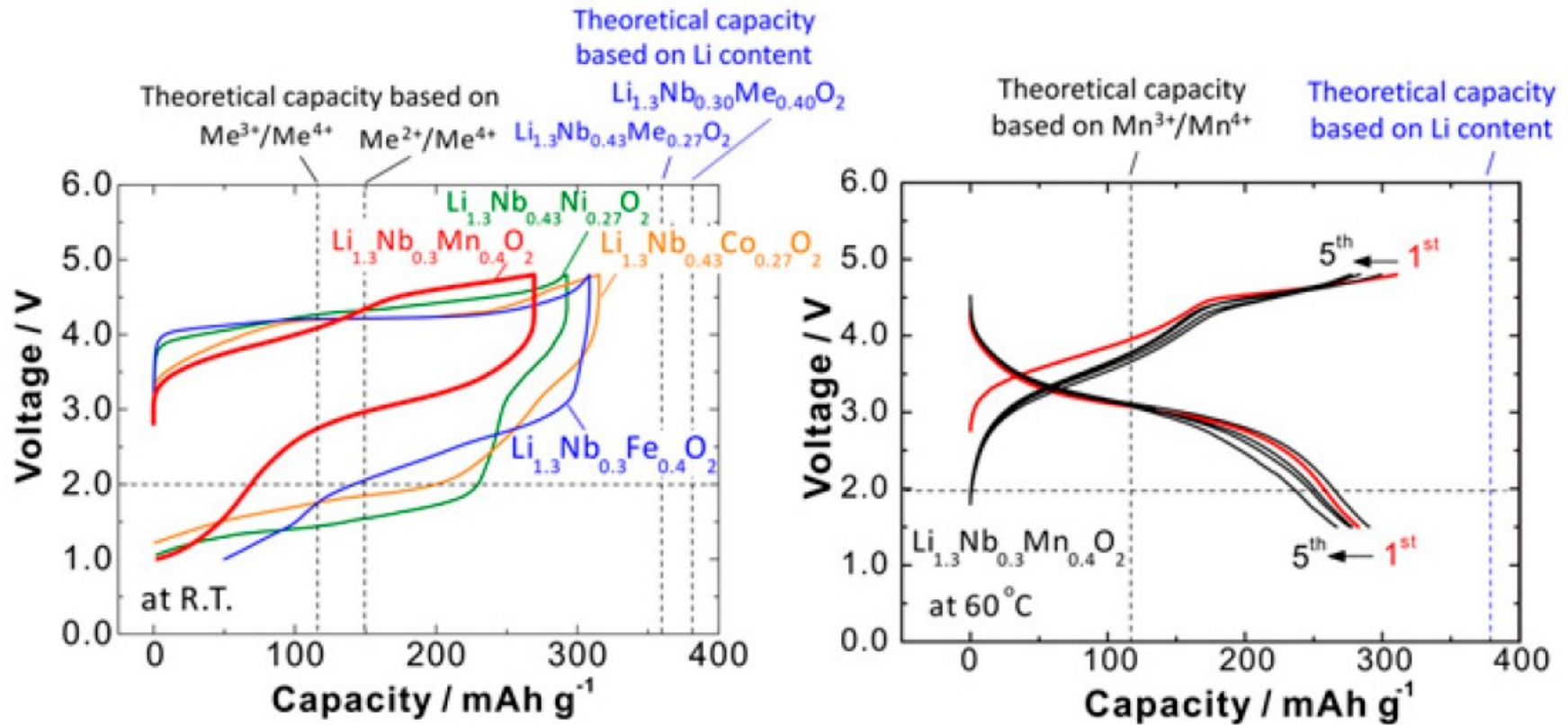
Irreversible capacity solely due to the oxygen evolution

partially charged  $\text{Li}_{3.27}\text{Fe}_{0.56}\text{TeO}_{5.5}$

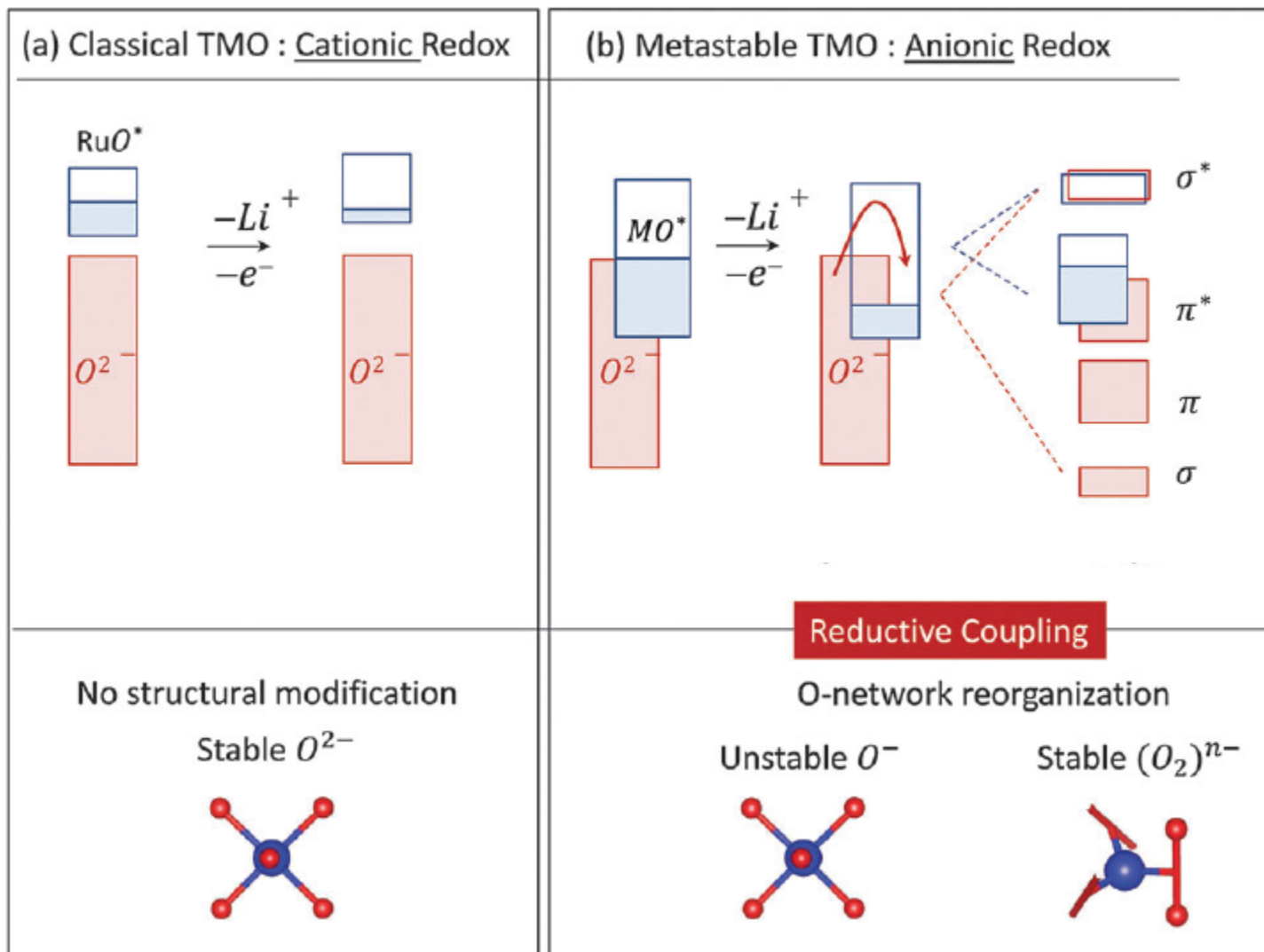


McCalla et al., JES, 162, A1341 (2015)

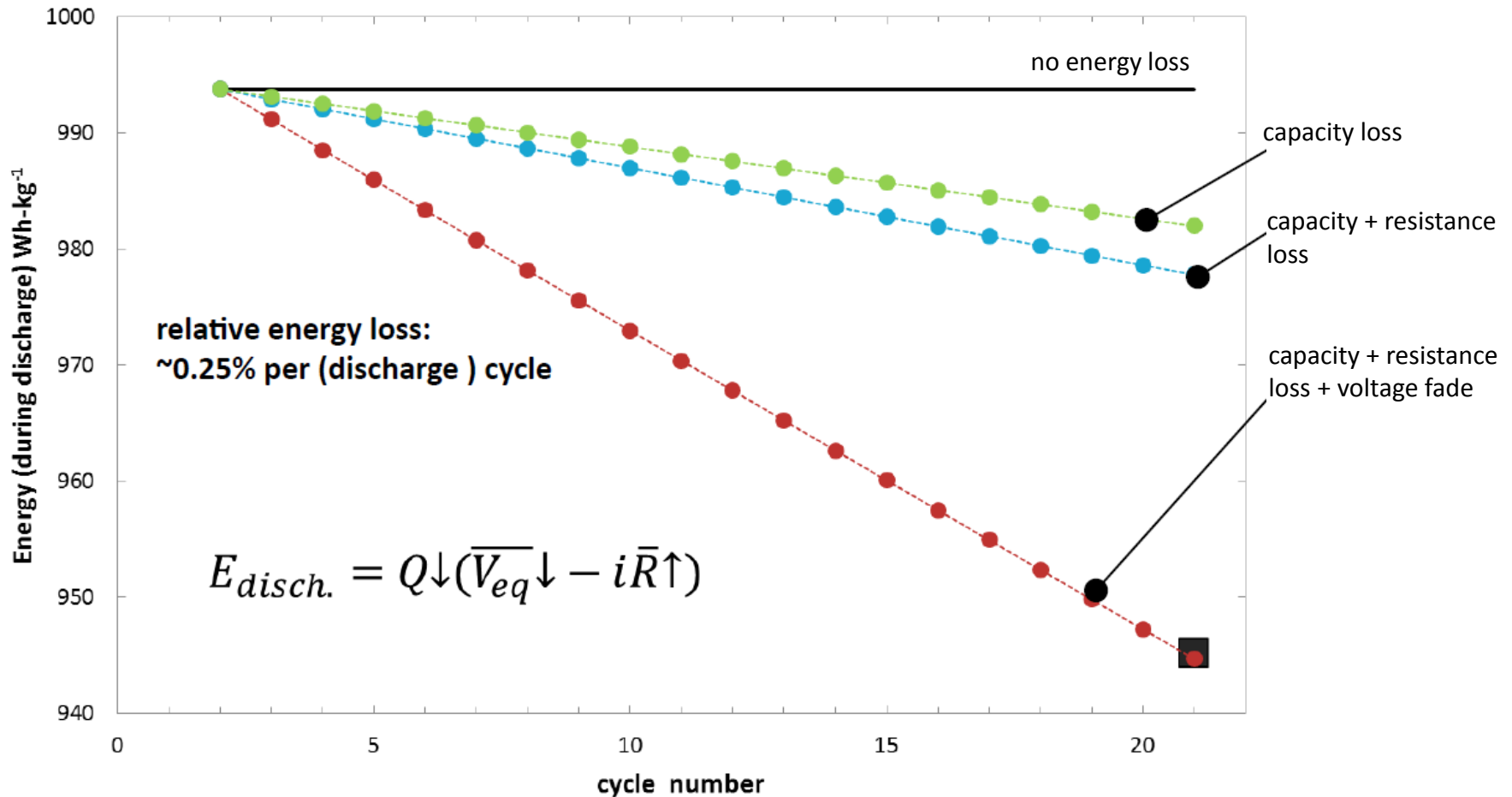
# Reversible oxygen oxidation



# Metastable anionic redox reaction



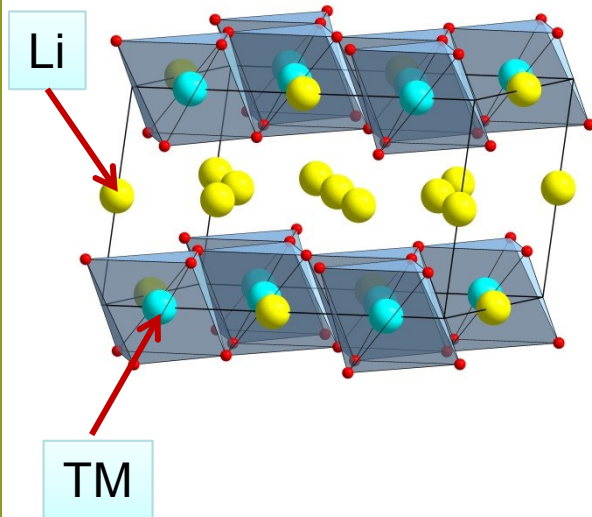
# High capacity layered cathodes: energy losses



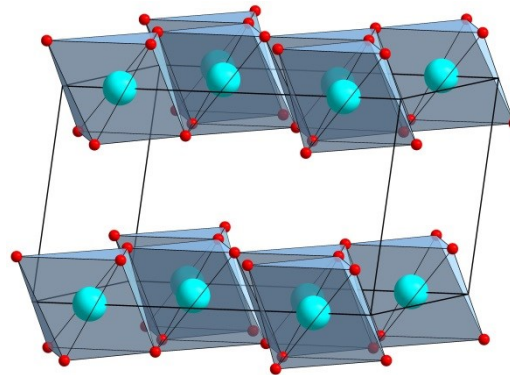
# High capacity layered cathodes: energy losses

- Do the TM cations migrate?
- What are the host positions?
- Is the migration reversible?

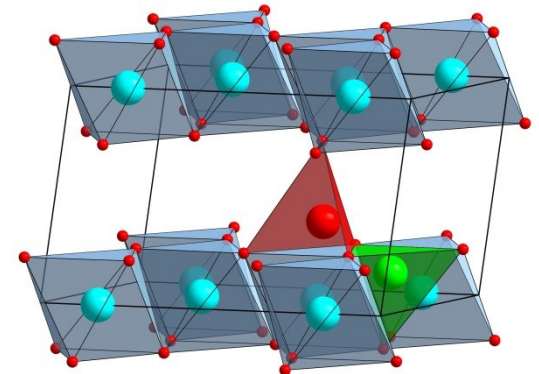
Pristine  $\text{Li}_2\text{MO}_3$   
M – 3d, 4d transition metal



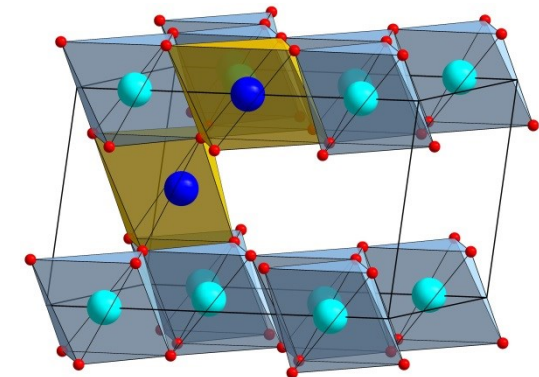
Hypothetic  
fully delithiated



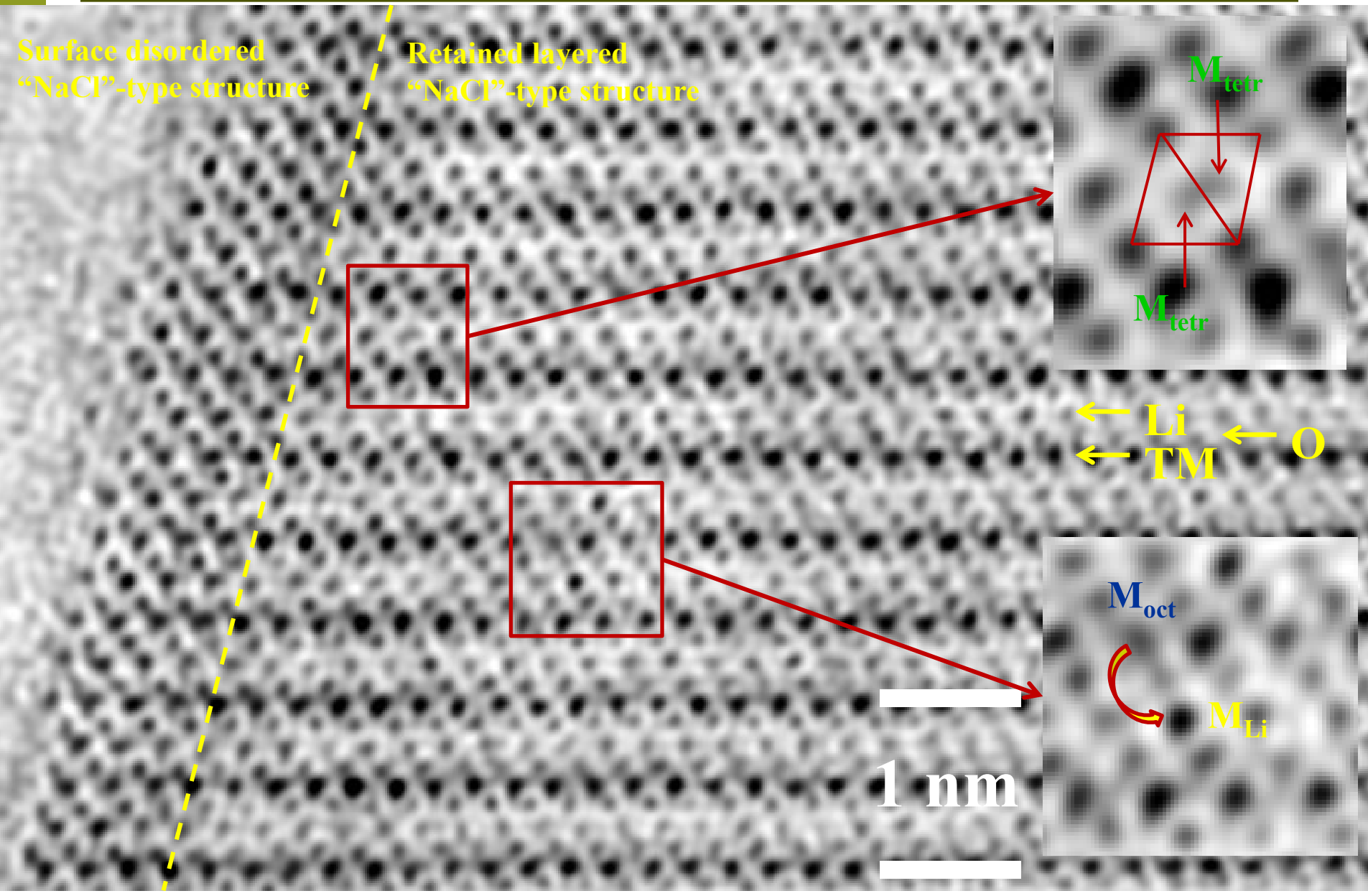
Migration of the M cations  
to intra- and interlayer  
octahedral sites



Migration of the M cations  
to intra- and interlayer  
tetrahedral sites

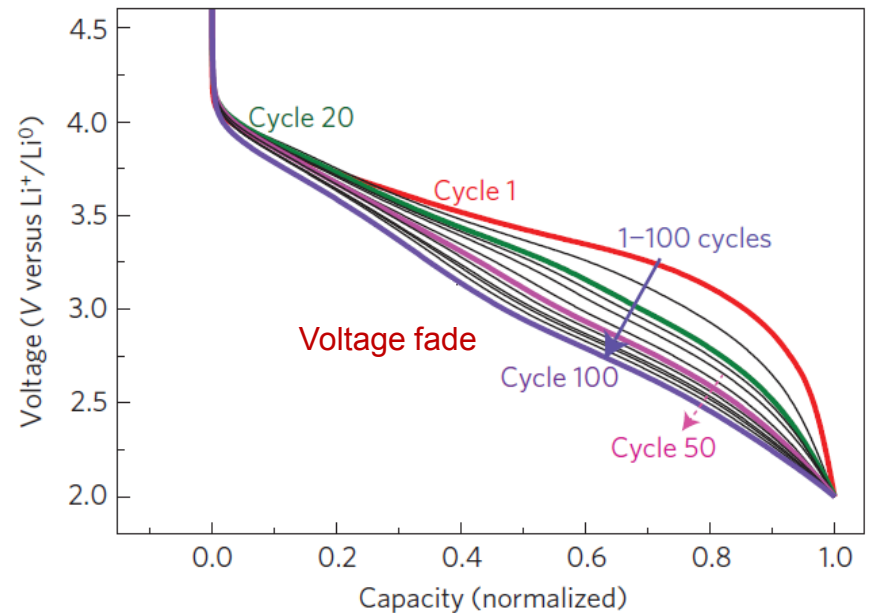
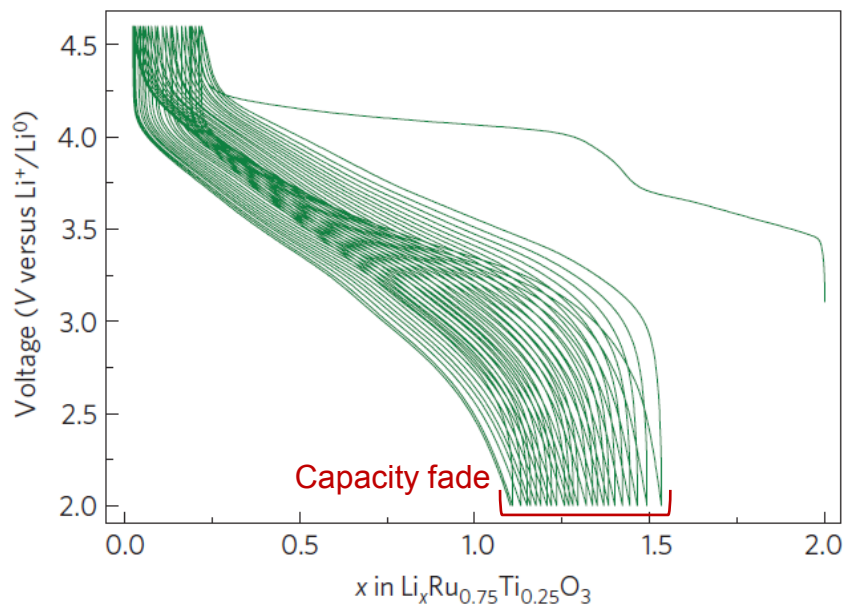


# High capacity layered cathodes: energy losses



# Capacity and voltage fade – $\text{Li}_2\text{Ru}_{0.75}\text{Ti}_{0.25}\text{O}_3$

## 100 charge/discharge cycles – discharge curves

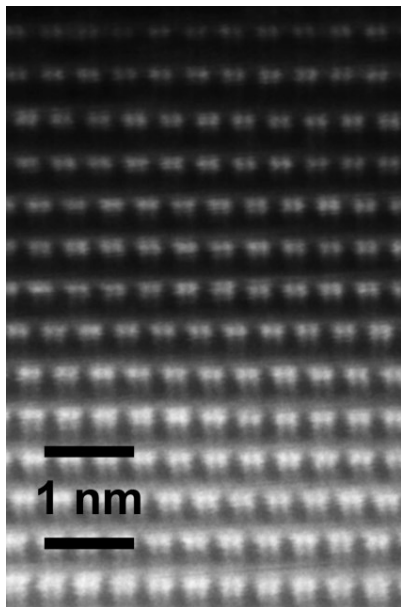


Sathiya, Abakumov, Foix, Rousse, Ramesha, Saubanère, Doublet, Vezin, Laisa, Prakash, Gonbeau, Van Tendeloo, Tarascon, *Nature Mater.*, 14, 230 2015

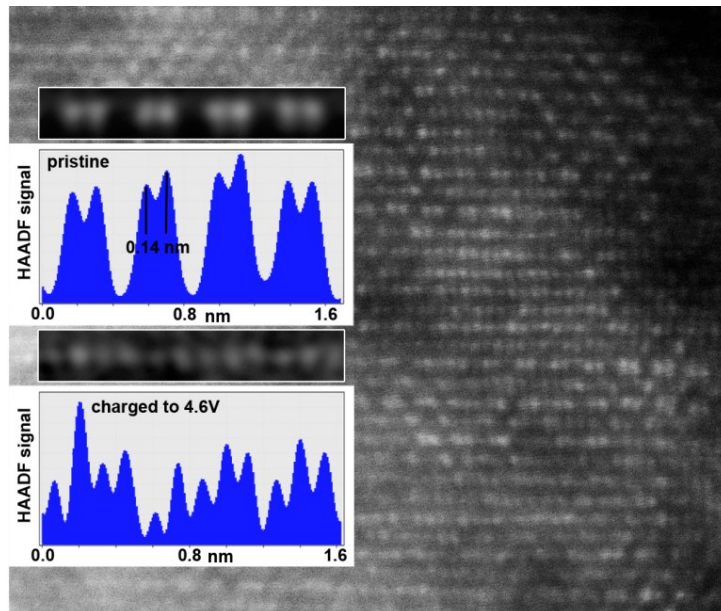


# TM cation migration – $\text{Li}_2\text{Ru}_{0.75}\text{Ti}_{0.25}\text{O}_3$

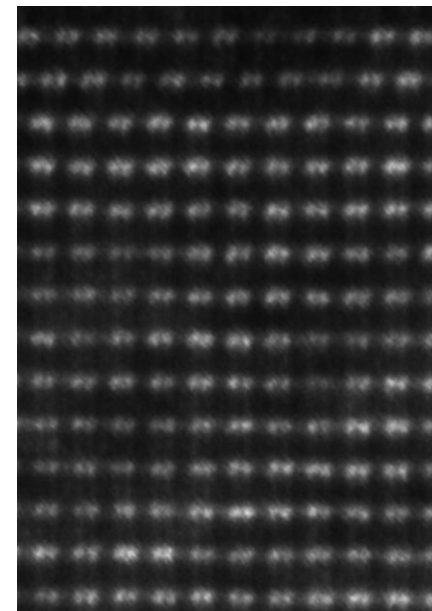
Pristine



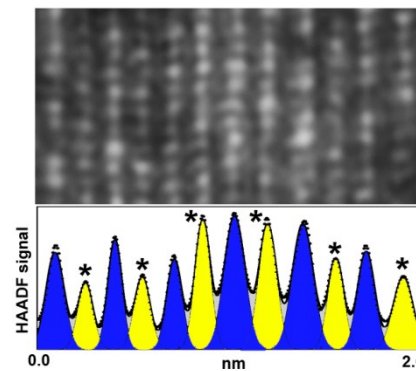
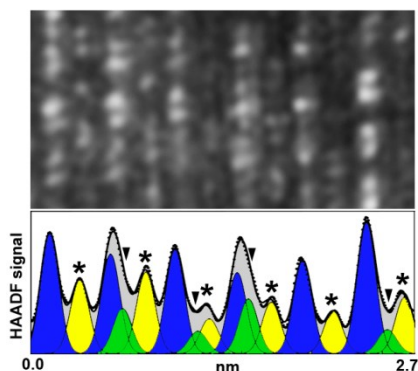
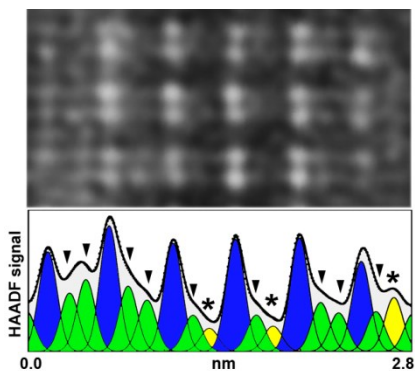
Charged to 4.6V



Discharged to 2V

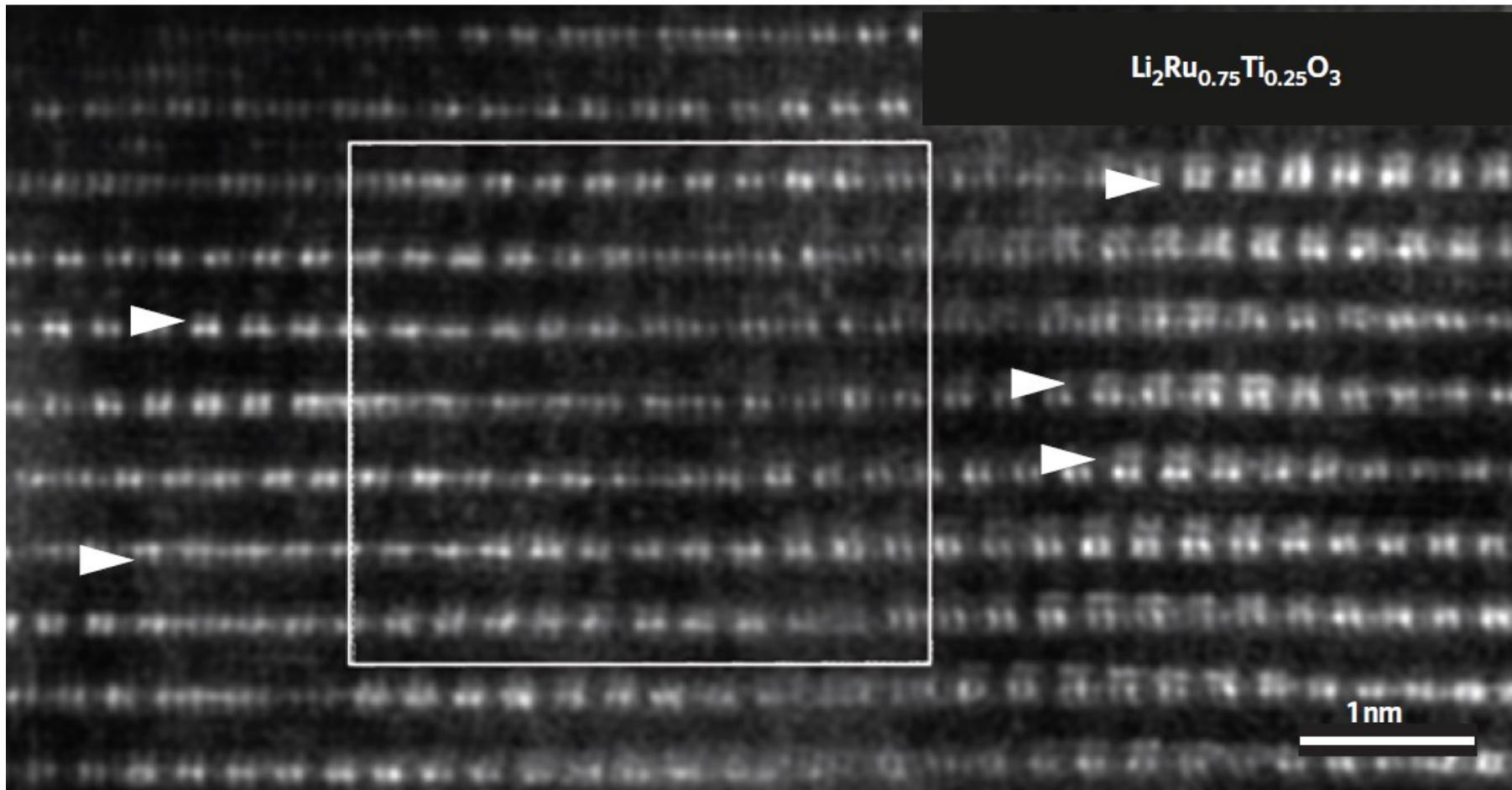


Structurally inhomogeneous charged state



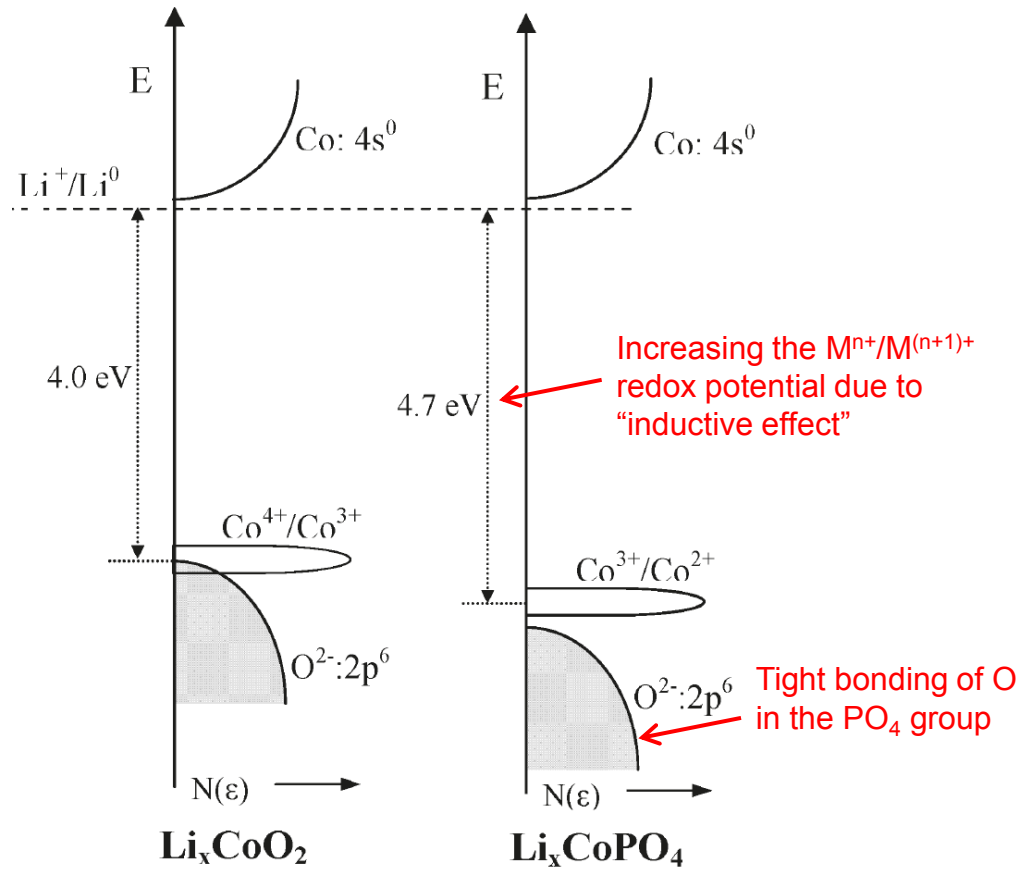
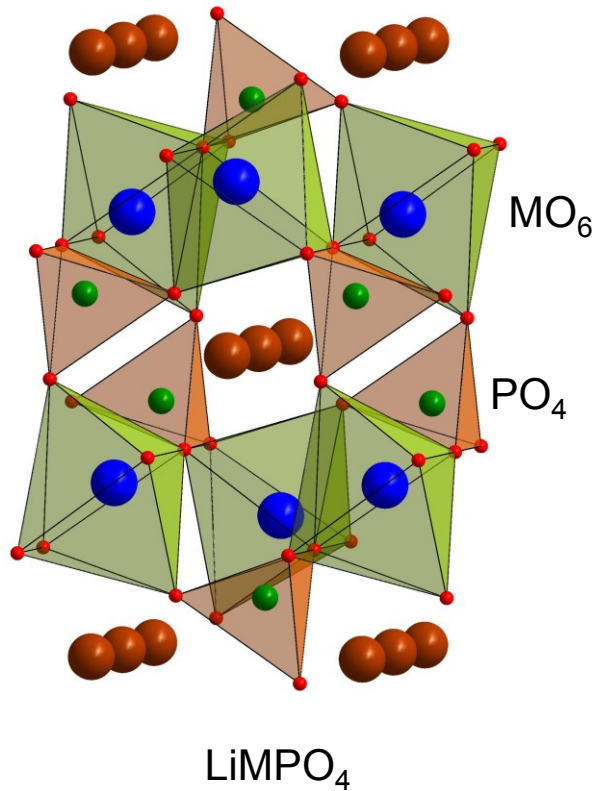
Sathiya, Abakumov, Foix, Rouse, Ramesha, Saubanère, Doublet, Vezin, Laisa, Prakash, Gonbeau, Van Tendeloo, Tarascon, *Nature Mater.*, 14, 230 2015

# TM cation migration – $\text{Li}_2\text{Ru}_{0.75}\text{Ti}_{0.25}\text{O}_3$

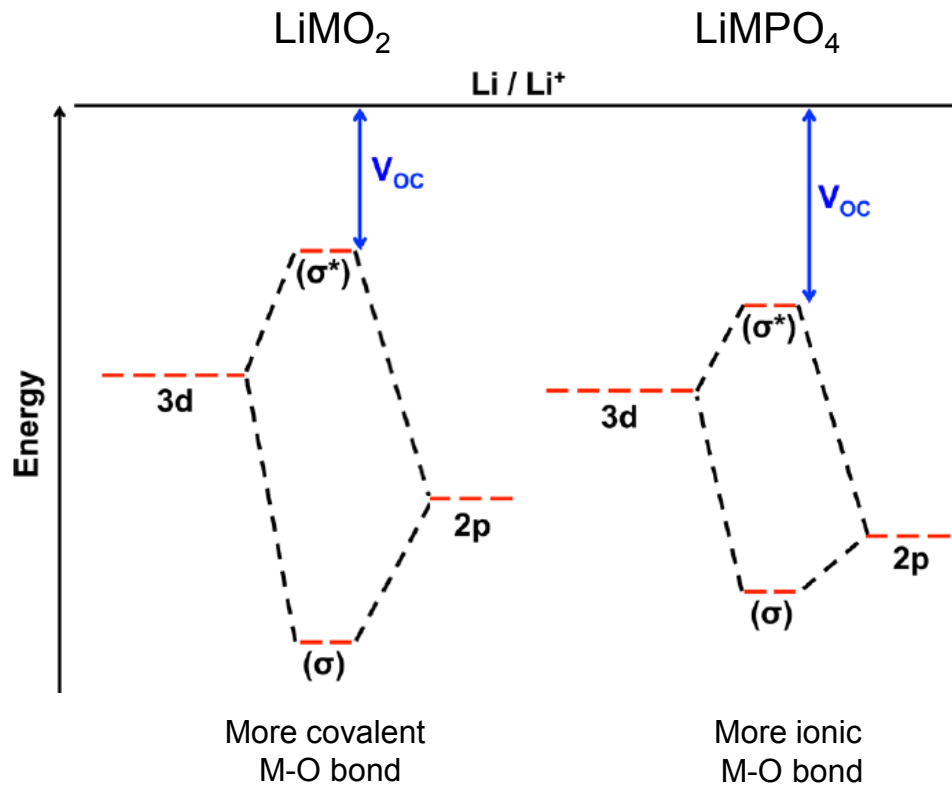


Discharged to 2V after 50 cycles: trapping of the TM cations at the tetrahedral interstices

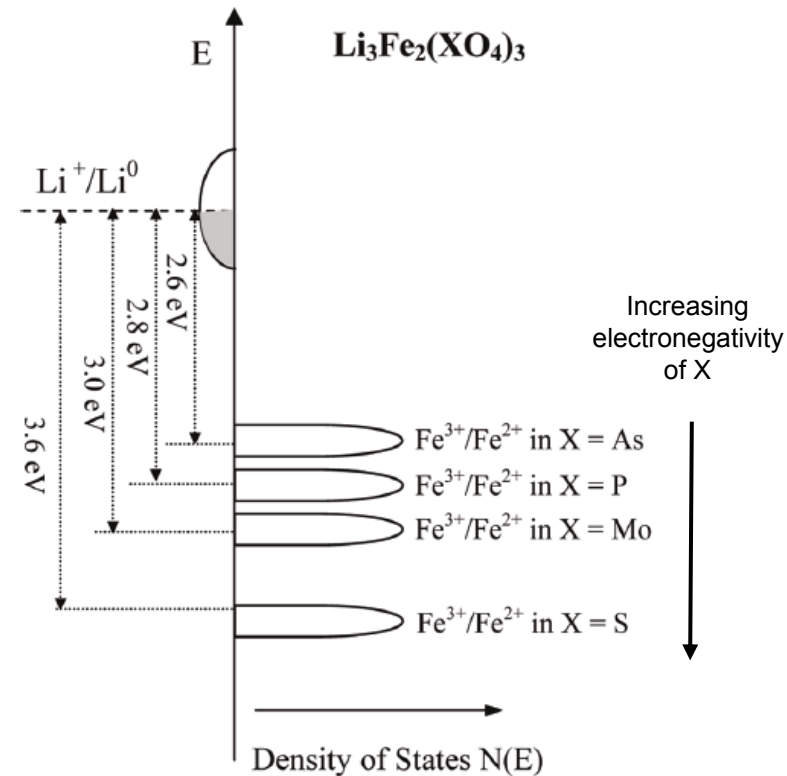
# Polyanion cathode materials



# Polyanion cathode materials



Tuning the M<sup>n+</sup>/M<sup>(n+1)+</sup> redox potential through adjusting the M-O-X interactions



Tuning the M<sup>n+</sup>/M<sup>(n+1)+</sup> redox potential through changing electronegativity of X

# Cathodes with the olivine structure: $\text{LiFePO}_4$

Capacity 170 mAh/g, voltage  $\sim 3.5$  V

## Pros:

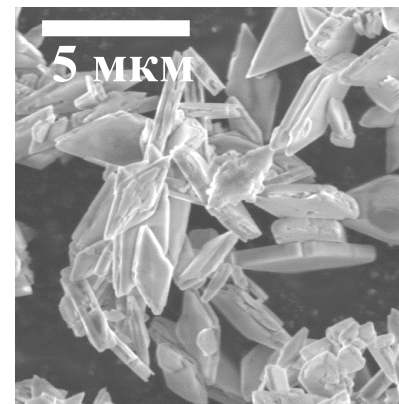
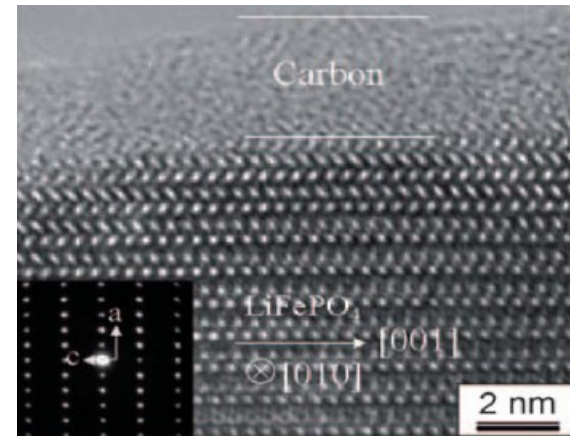
- stability (3D structure +  $\text{PO}_4$ )
- $\text{LiFePO}_4 \leftrightarrow \text{FePO}_4 + \text{Li}^+ + \text{e}^-$
- environmentally benign
- low cost

## Solution:

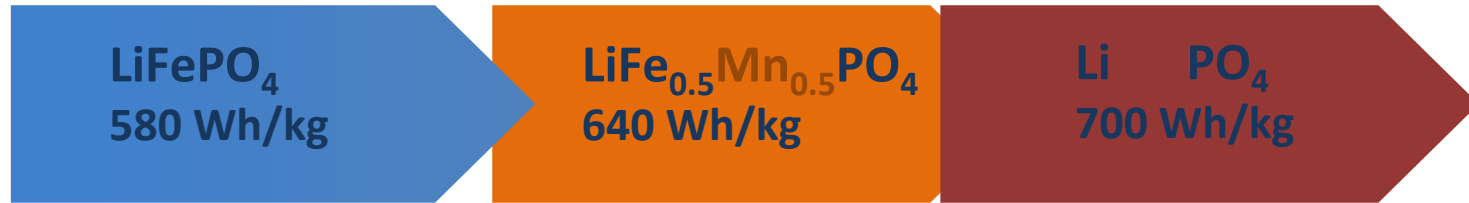
- conducting carbon coating
- nanosized particles
- optimized morphology
  
- platelets with 200 – 300 nm thickness along the fast diffusion direction
- high discharge current (50% of discharge capacity / 1 min)

## Cons:

- low conductivity  $\sim 10^{-9}$  S/cm
- low  $\text{Li}^+$  diffusion coefficient  $\sim 10^{-15}$   $\text{cm}^2/\text{s}$
- relatively low voltage

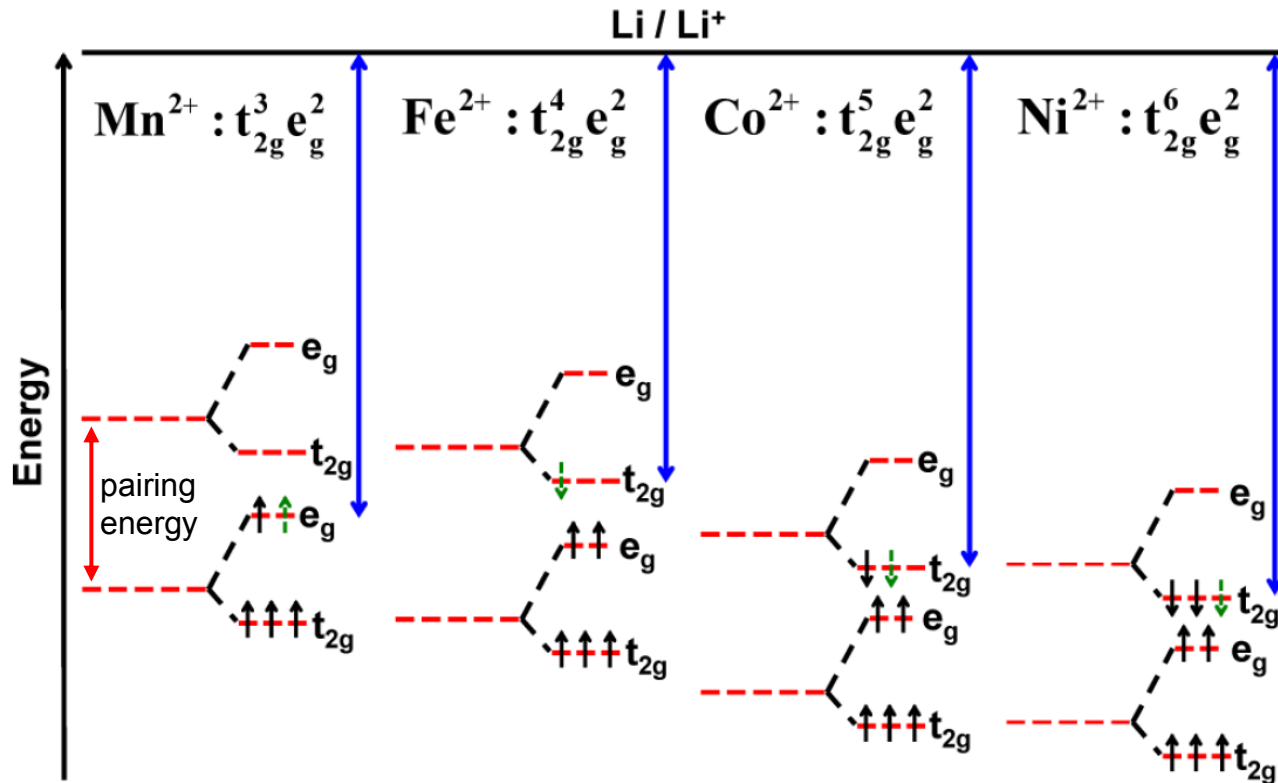


# Cathodes with the olivine structure: $\text{Li}(\text{Mn,Fe})\text{PO}_4$

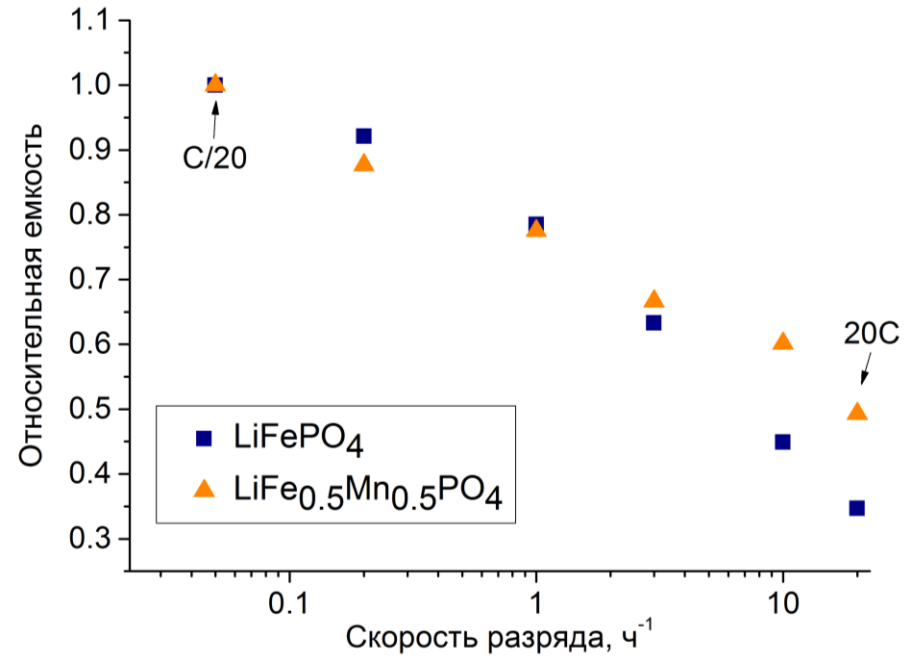
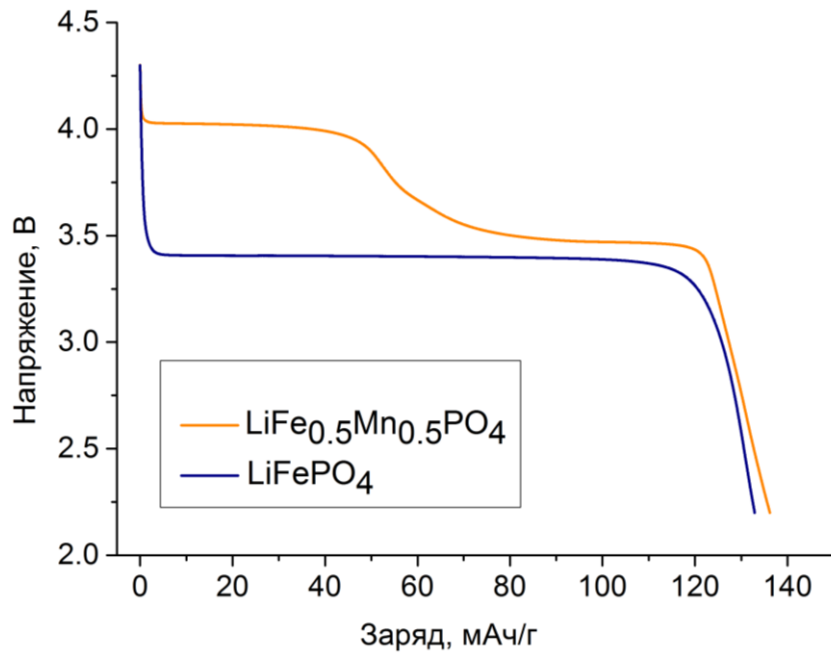


battery with  $\approx 120 - 140$  Wh/kg

battery up to 150 - 175 Wh/kg



# Cathodes with the olivine structure: $\text{Li}(\text{Mn,Fe})\text{PO}_4$

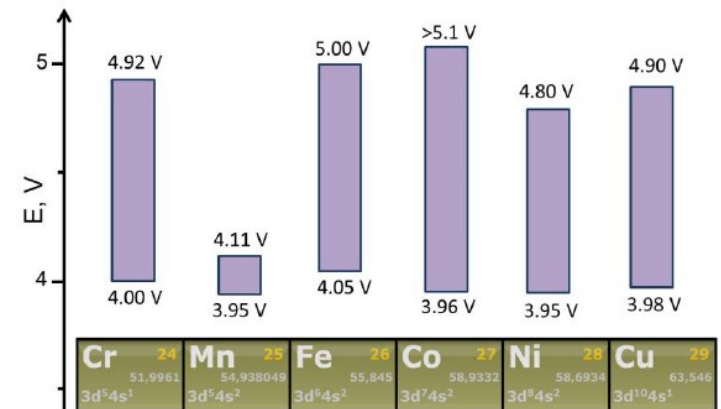
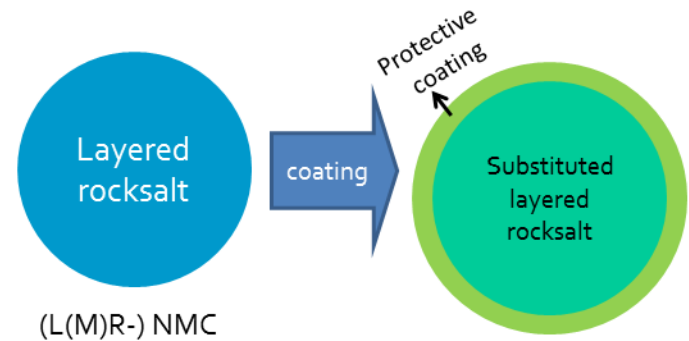


O.Drozhzhin, V.Sumanov, O.Karakulina, A.Abakumov, J.Hadermann, A.Baranov, K.Stevenson, E.Antipov,  
*J.Power Sources*, 2015

# Conclusions

## Improvement of the cathode materials

1. Chemical substitutions (metals with strong covalent bonding to O, cations blocking migration)
2. Protective coatings
3. Maximizing the energy density through tuning redox potential
4. Nanostructuring and functional coatings
5. New crystal structures and chemistries



Stability windows for the  $\text{LiMn}_{1.5}\text{M}_{0.5}\text{O}_4$  spinels

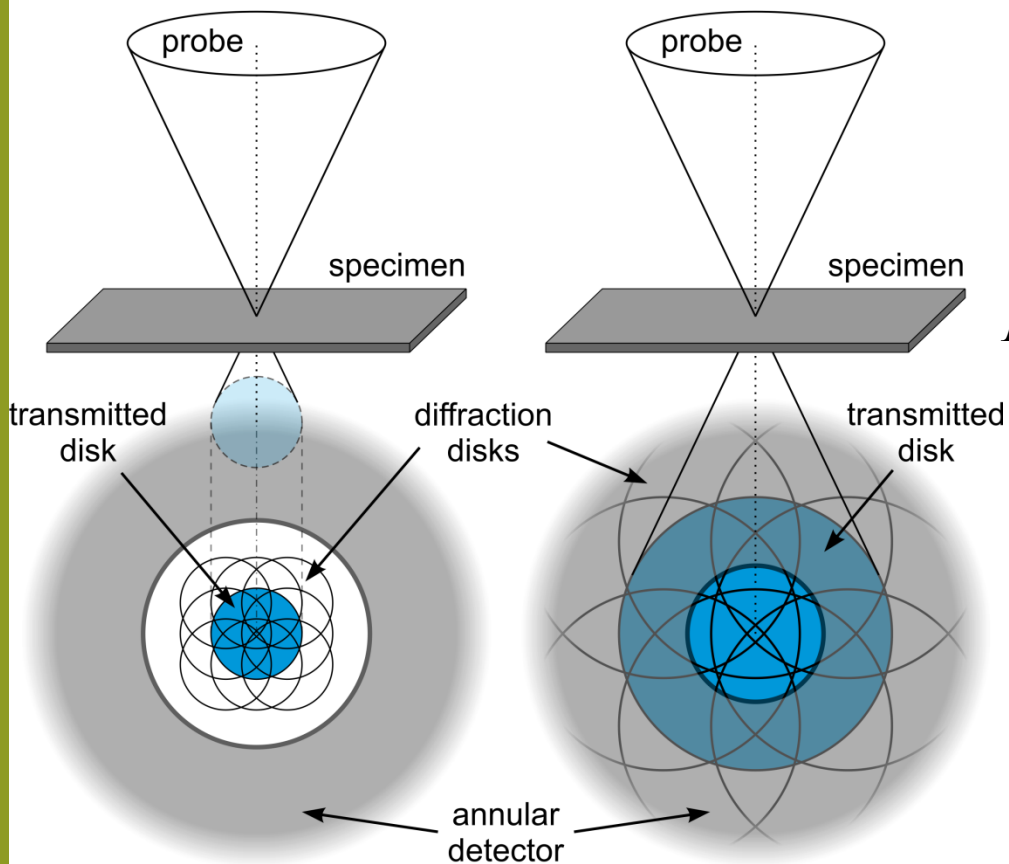


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**Thank you for your attention!**

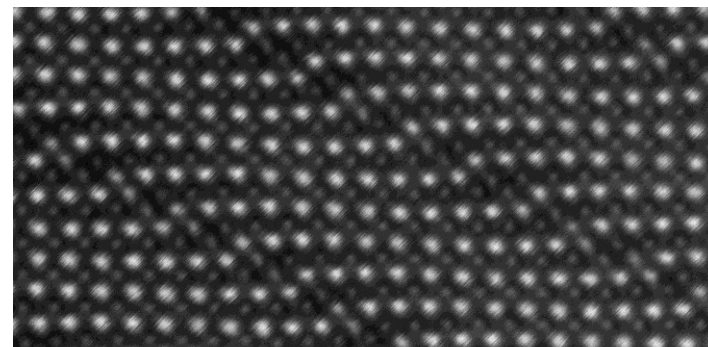
## HAADF-STEM

## ABF-STEM

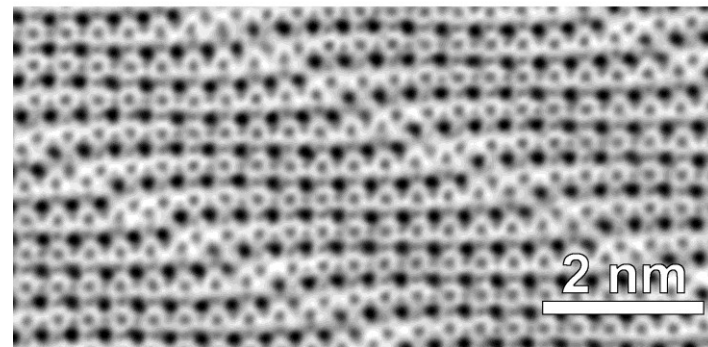


$$I \sim Z^2$$

$$I \sim Z^{1/3}$$



HAADF-STEM



ABF-STEM

**HAADF-STEM** – high angle annular dark field scanning transmission electron microscopy

**ABF-STEM** – annular bright field scanning transmission electron microscopy